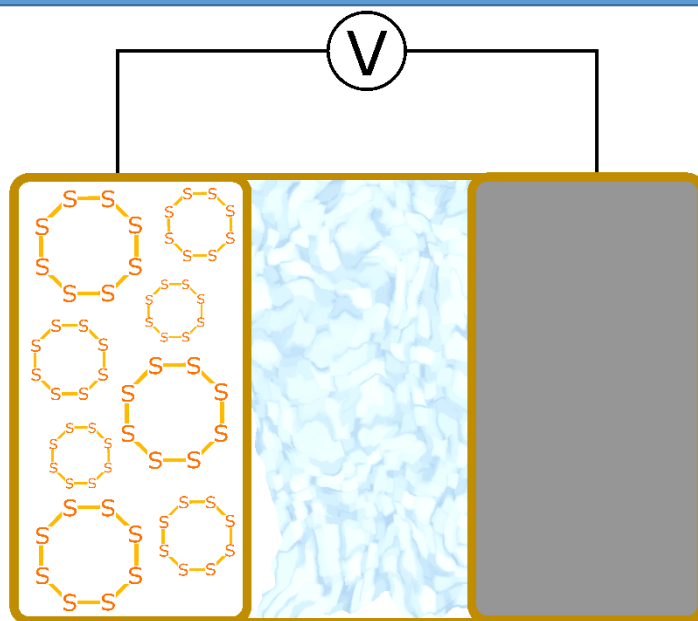


# Lithium–Sulfur batteries and discharge products from cycling

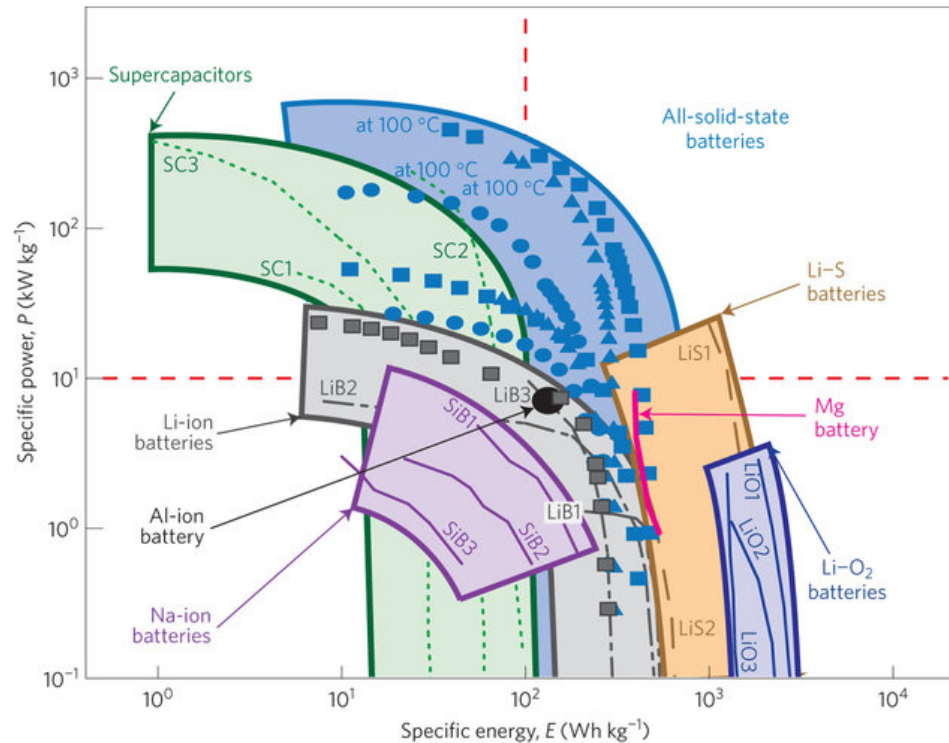


Molleigh Preefer

MTRL 286G, Spring 2016

See, K. A.; Leskes, M.; Griffin, J. M.; Britto, S.; Matthews, P. D.; Emly, A.; Van der Ven, A.; Wright, D.S.; Morris, A.J.; Grey, C.P.; Seshadri, R. **Ab Initio Structure Search and in Situ  $^7\text{Li}$  NMR Studies of Discharge Products in the Li–S Battery System.** *Journal of the American Chemical Society*, 2014, 136(46), 16368-16377.

# Why Li-S batteries?



- |  |  |  |
|--|--|--|
| ■ $\text{Li}_4\text{Ti}_5\text{O}_{12}/\text{LiNbO}_3$ -coated $\text{LiCoO}_2$  | SC1: activated carbon/activated carbon                               | LiB1: graphite/ $\text{LiCoO}_2$   |
| ● $\text{Li}_4\text{Ti}_5\text{O}_{12} + \text{LGPS}/\text{LGPS}/\text{LiNbO}_3$ -coated $\text{LiCoO}_2 + \text{LGPS}$          | SC2: reduced graphene oxide $\text{RuO}_2/\text{RuO}_2$ -polyaniline | LiB2: $\text{Li}/\text{LiFePO}_4$  |
| ▲ $\text{Li}_4\text{Ti}_5\text{O}_{12} + \text{LSiPSCl}/\text{LSiPSCl}/\text{LiNbO}_3$ -coated $\text{LiCoO}_2 + \text{LSiPSCl}$ | SC3: activated carbon/activated carbon                               | LiB3: $\text{Li}_4\text{Ti}_5\text{O}_{12}/\text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_4$ |
| ■ Graphite + LPS/LPS   LGPS/ $\text{LiNbO}_3$ -coated $\text{LiCoO}_2 + \text{LGPS}$   |  |  |
| SiB1: $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ (NVP) + graphene/NVP + graphene   | LiS1: $\text{Li}/\text{S}$ (graphene + single-walled CNT)            | LiO1: $\text{Li}/\text{O}_2$ (graphene)  |
| SiB2: NVP + CNT/NVP + CNT  | LiS2: $\text{Li}/\text{S}$   | LiO2: $\text{Li}/\text{O}_2$ (carbon nanofibres)                                       |
| SiB3: NVP + activated carbon (AC)/NVP + AC   |  | LiO3: $\text{Li}/\text{O}_2$ (carbon nanotubes (CNT))                                  |
| Al-ion battery: Al/graphite  |  |  |
| Mg battery: $\text{Mg}/\text{V}_2\text{O}_5$   |  |  |

Kato, Y.; Hori, S.; Saito, T.; Suzuki, K.; Hirayama, M.; Mitsui, A.; Yonemura, M.; Iba, H.; Kanno, R. High-power all-solid-state batteries using sulfide superionic conductors. *Nature Energy*, 2016, 1, 16030.

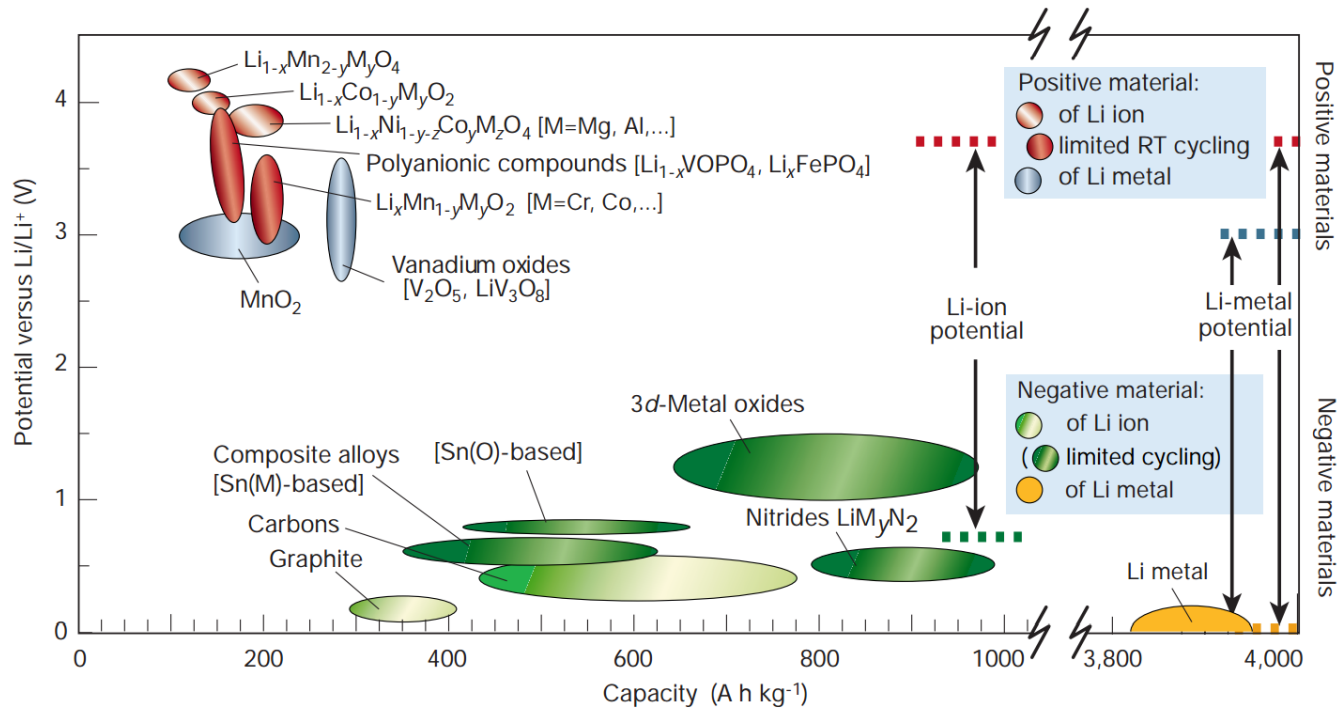
# Cathode: why sulfur?



Recovered sulfur as an industrial side product of oil refining in British Columbia

- Inexpensive
- Abundant
- Non-toxic
- High theoretical capacity ( $1672 \text{ mAhg}^{-1}$ )

# Anode: why lithium?



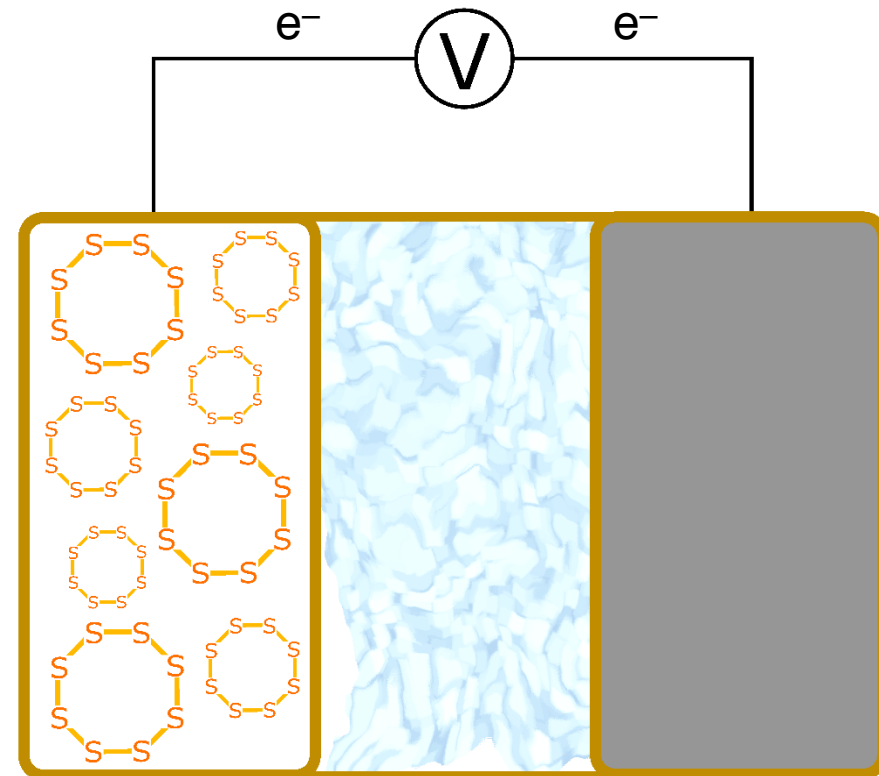
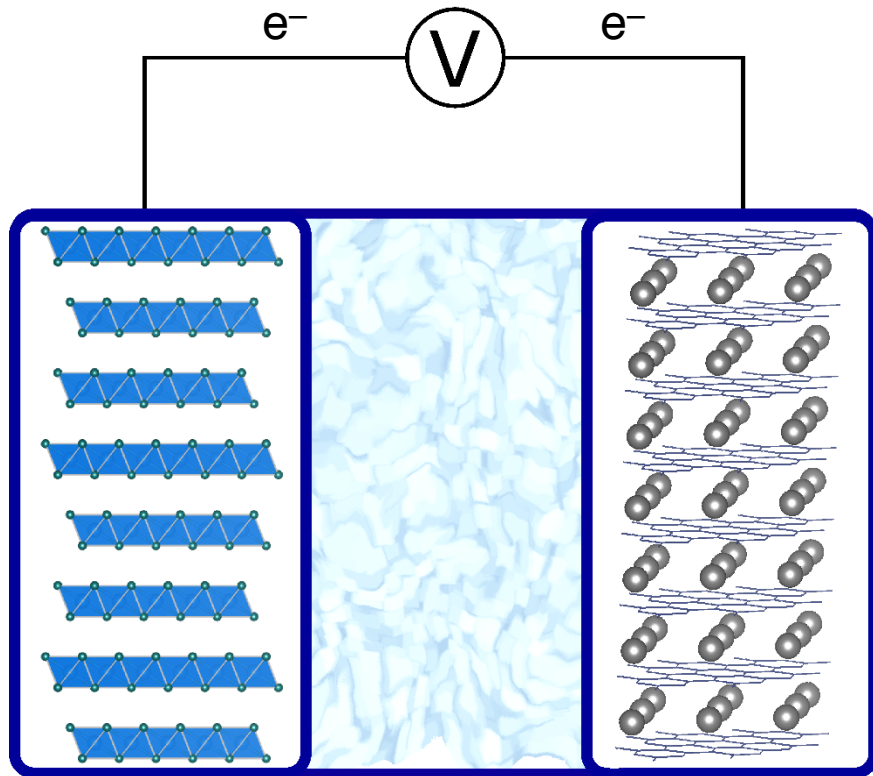
Graphite (typically used in Li-ion batteries) has a theoretical specific capacity of  $372 \text{ mAhg}^{-1}$  (operating capacity of  $\sim 350 \text{ mAhg}^{-1}$ ) which is incompatible with a sulfur cathode

Li metal has a theoretical specific capacity of  $3860 \text{ mAhg}^{-1}$

# The difference between Li-S and Li-ion

Li-ion

Li-sulfur

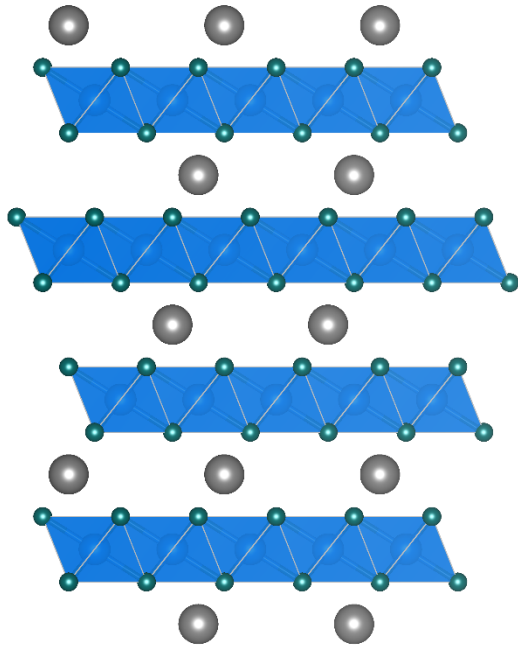


Cathode:  $\text{Li}_{1-x}\text{CoO}_2$   
Anode:  $\text{Li}_x\text{C}_6$

Cathode: Sulfur  
Anode: Lithium metal

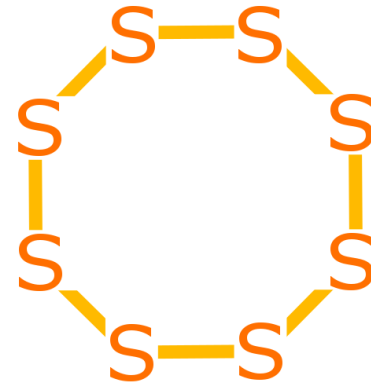
# Charged cathode structures

Li-ion cathode



Cathode:  $\text{Li}_{1-x}\text{CoO}_2$   
Anode:  $\text{Li}_x\text{C}_6$

Li-sulfur cathode



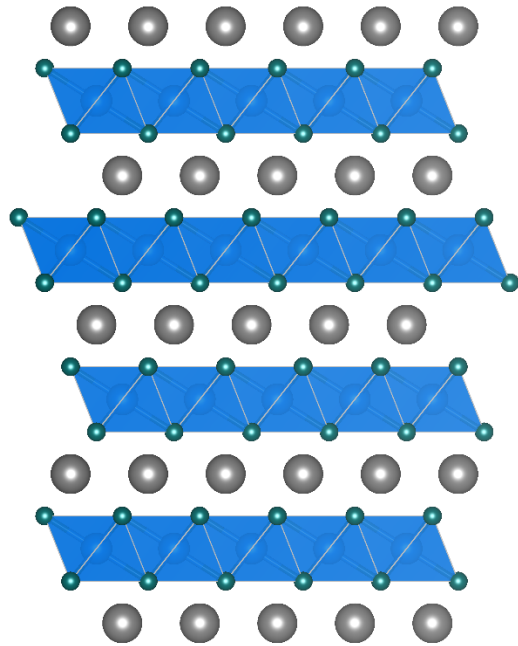
Cathode: Sulfur  
Anode: Lithium metal

**charge**

delithiate  
oxidize

# Discharged cathode structures

Li-ion cathode

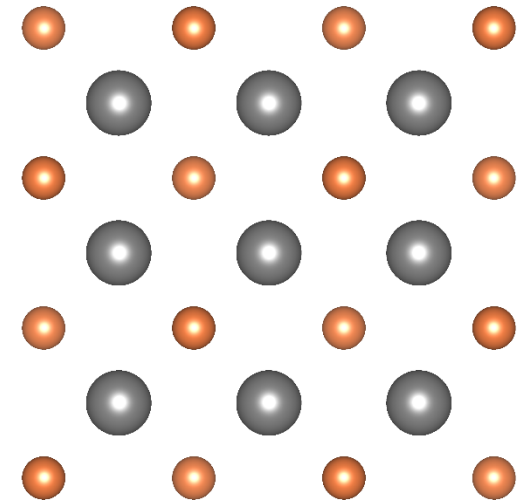


Cathode:  $\text{Li}_{1-x}\text{CoO}_2$   
Anode:  $\text{Li}_x\text{C}_6$

**discharge**

lithiate  
reduce

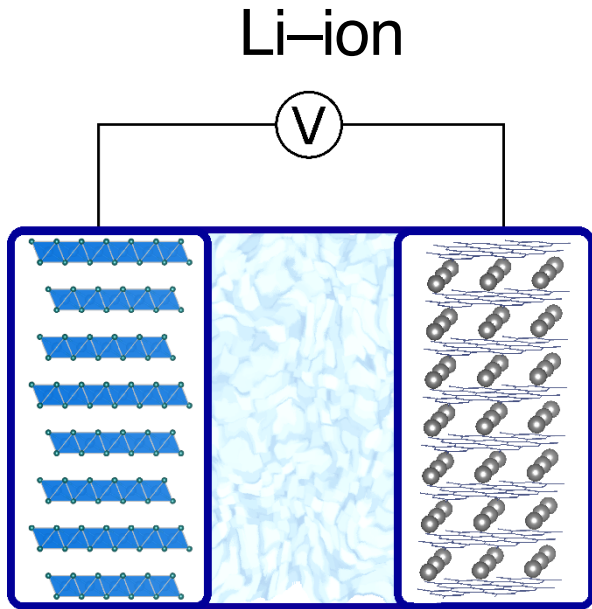
Li-sulfur cathode



Cathode: Sulfur  
Anode: Lithium metal

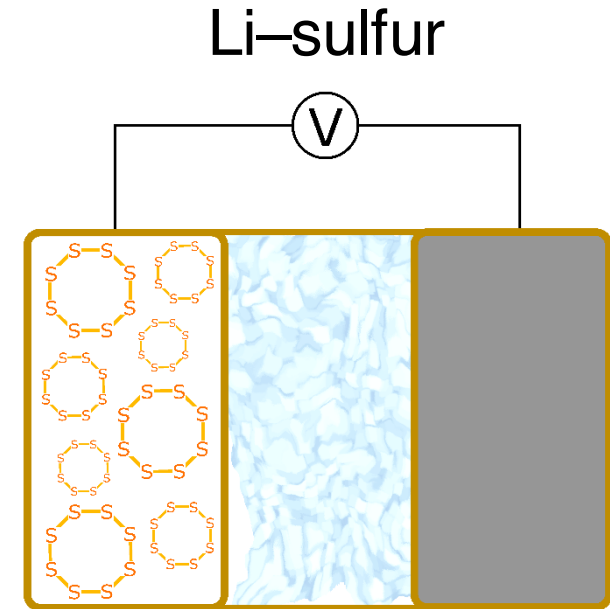


# Capacity of Li-S vs Li-ion



Theoretical capacity  $\text{LiCoO}_2$ :  
 **$274 \text{ mAhg}^{-1}$**

Structural limitation of capacity:  
 $\sim 150 \text{ mAhg}^{-1}$



Theoretical capacity:

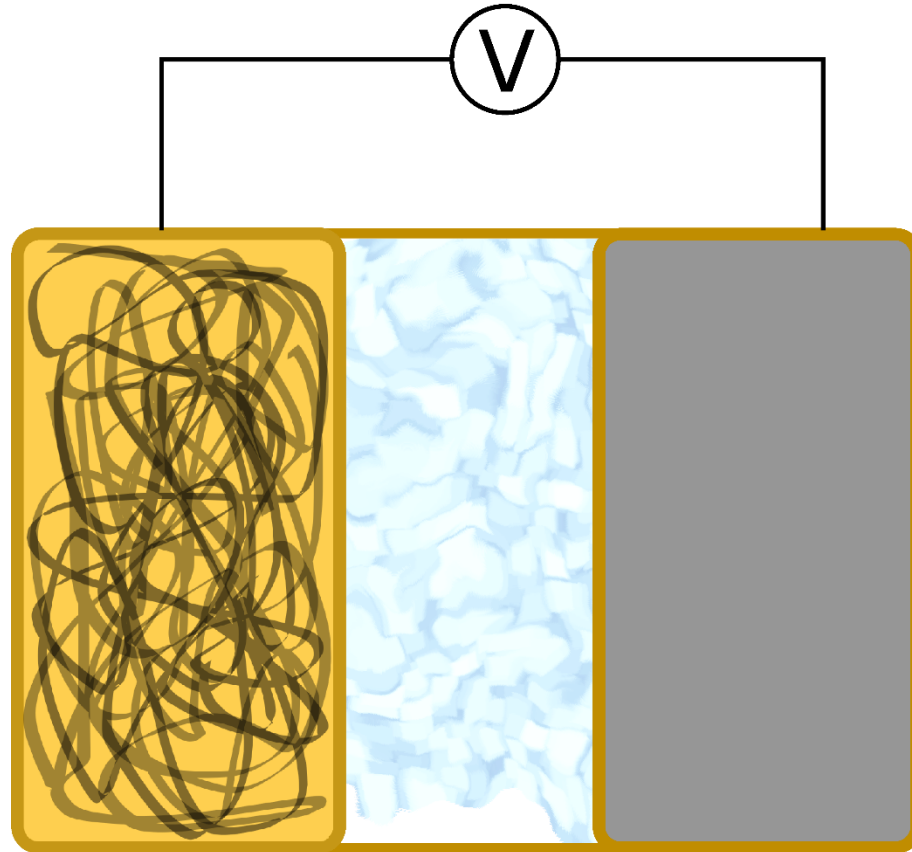
$$\frac{2 \text{ mol } e^-}{1 \text{ mol S}} \cdot \frac{96485 \frac{\text{C}}{\text{mol } e^-}}{32.06 \frac{\text{g}}{\text{mol S}}} \cdot \frac{1 \text{ mAh}}{3.6 \text{ C}}$$

**$=1672 \text{ mAhg}^{-1}$**



# Li-S challenges

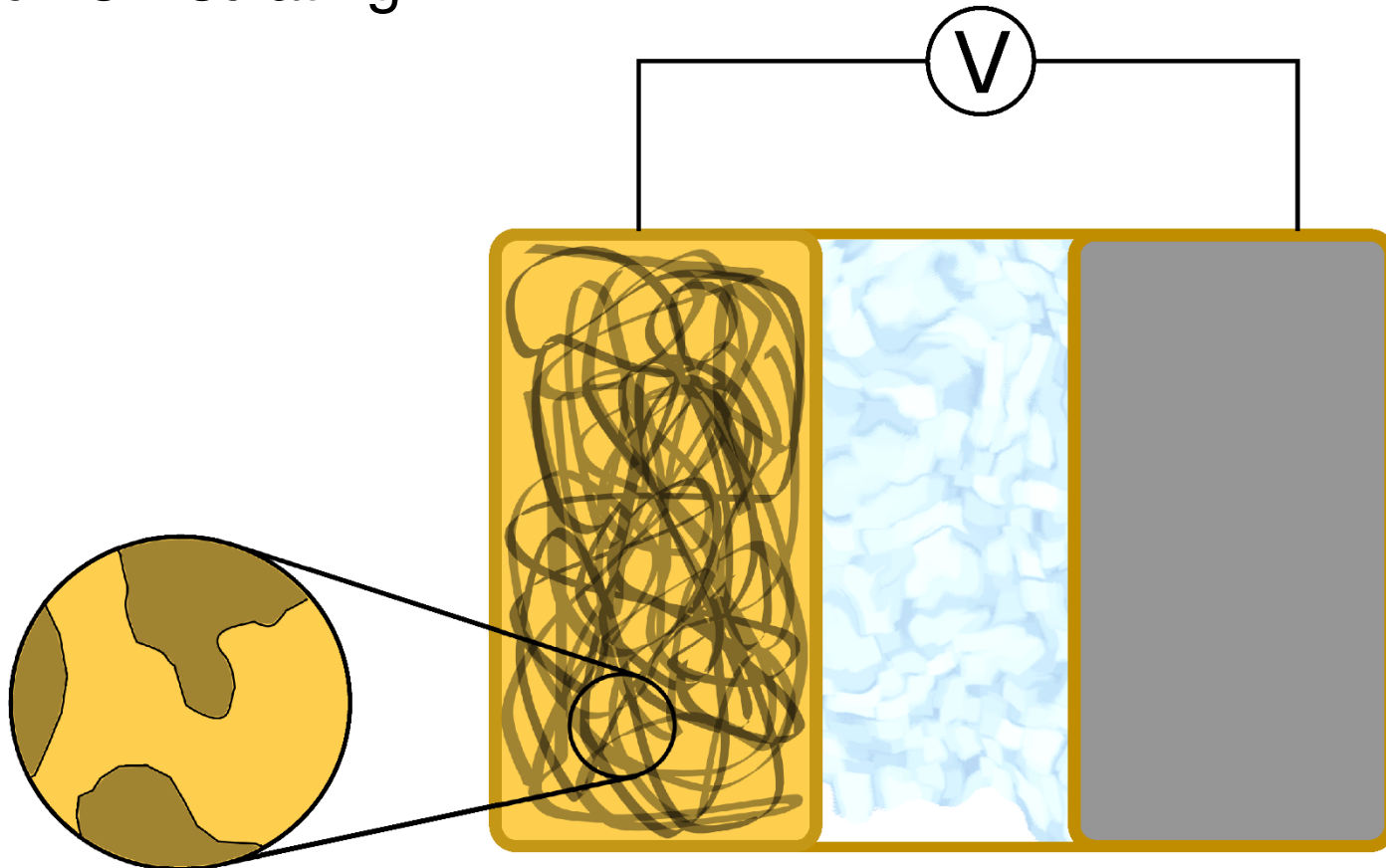
## 1. Sulfur is insulating



The cathode must include a mixture of conducting carbon and sulfur to get sufficient charge and ion mobility needed for battery cycling

# Li-S challenges

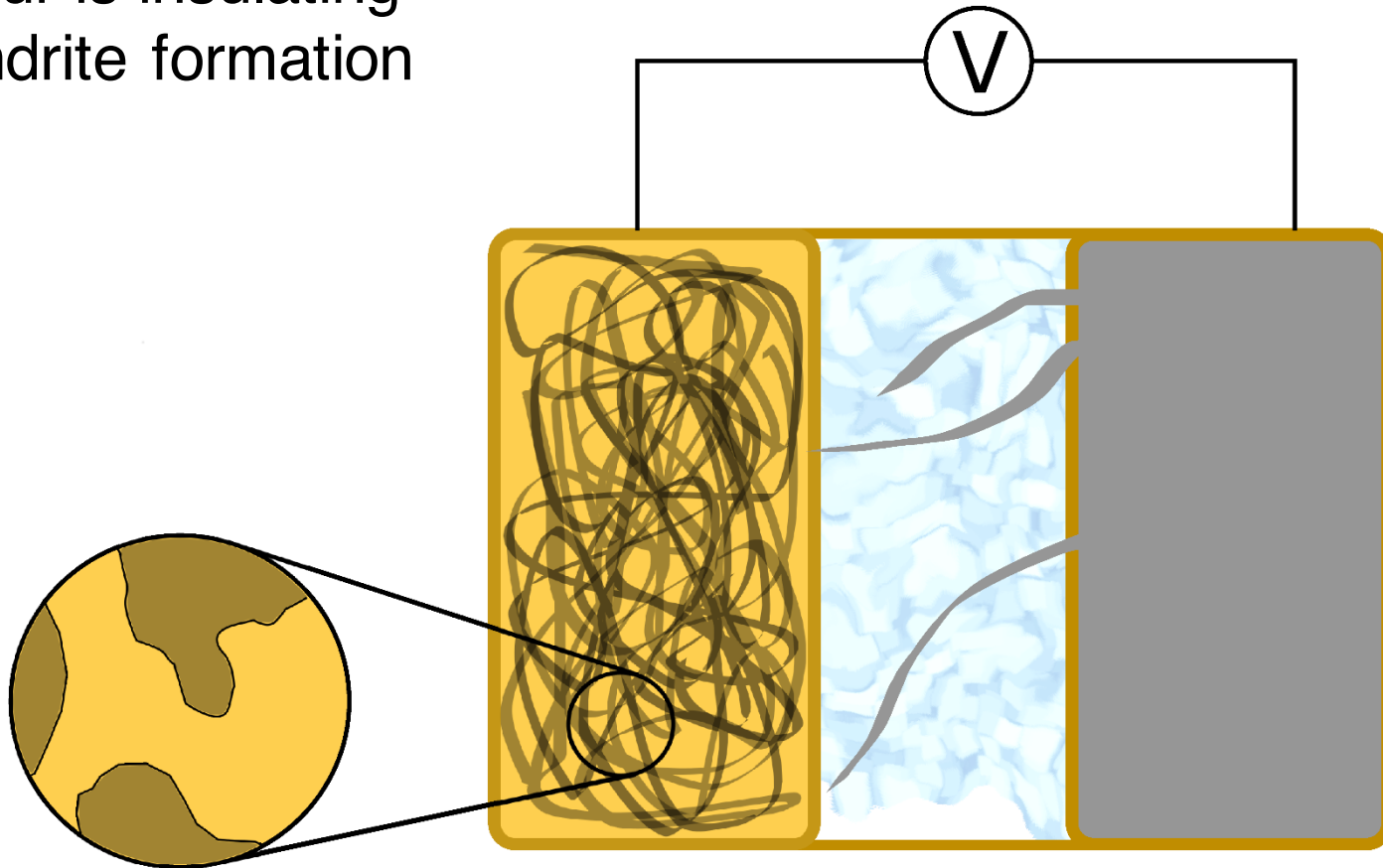
## 1. Sulfur is insulating



Adding conducting carbon presents an engineering problem to get around the issue of having distinct domains of carbon and sulfur

# Li-S challenges

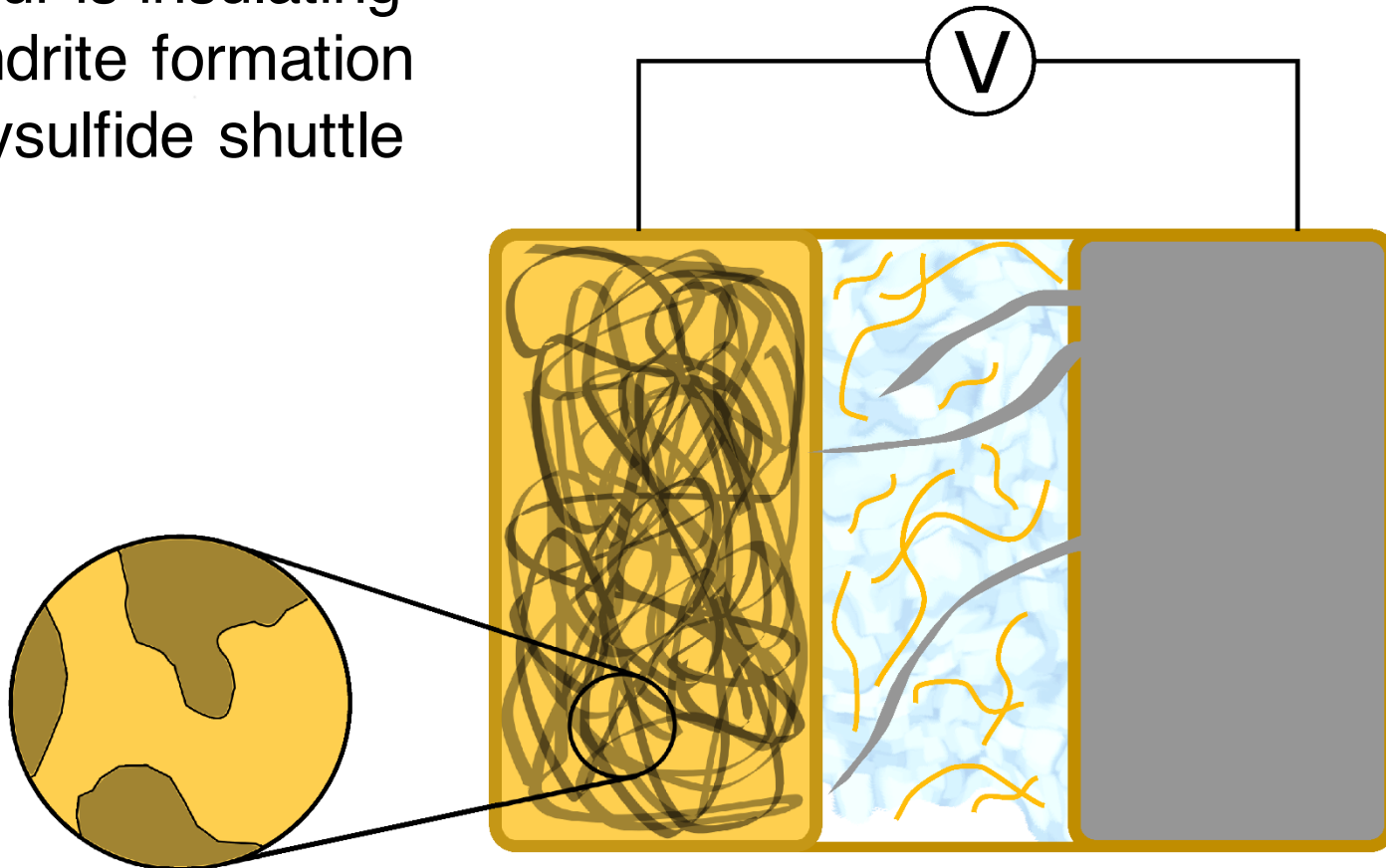
1. Sulfur is insulating
2. Dendrite formation



Dendrites can be atomically thin fibers of Li that extend from the anode to the cathode over many cycles and cause short-circuiting + safety concerns

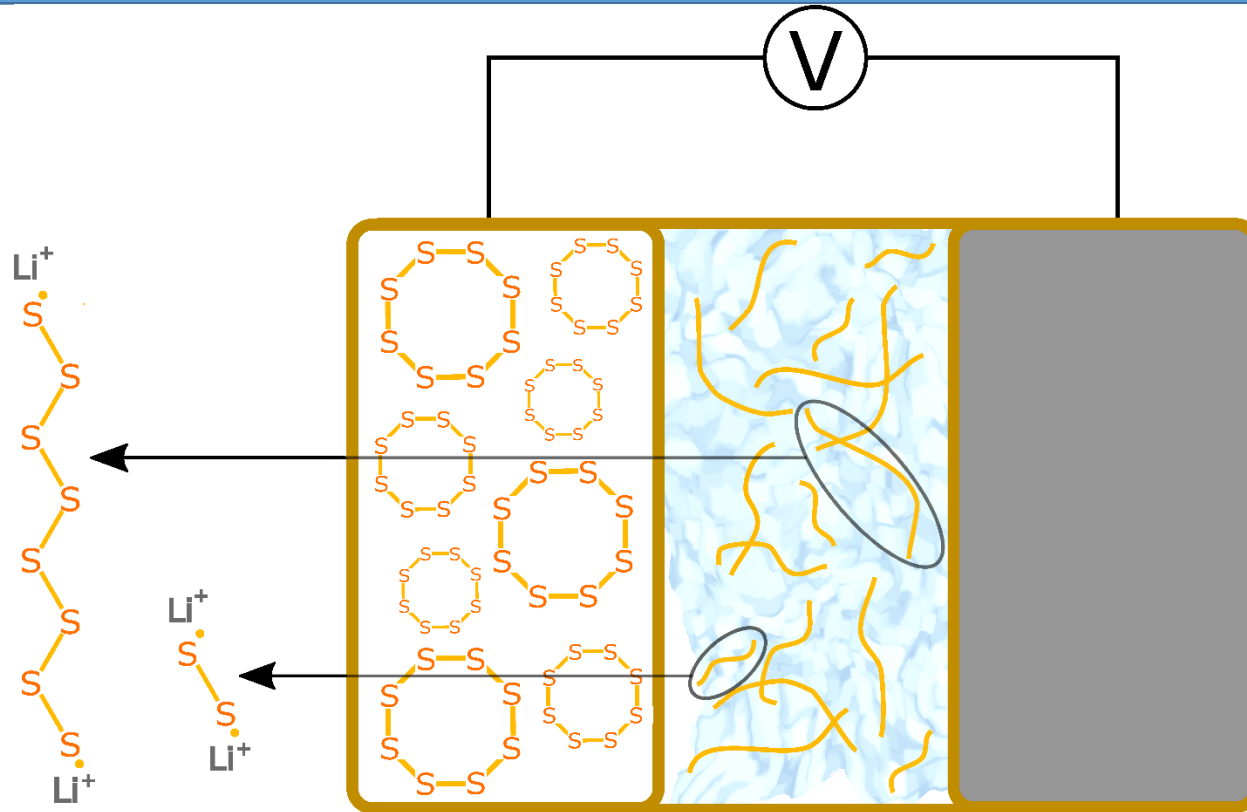
# Li-S challenges

1. Sulfur is insulating
2. Dendrite formation
3. Polysulfide shuttle



Chains of diradical sulfur and their counter lithium cations solubilize in the liquid electrolyte during discharge, causing capacity fade

# Polysulfide chains participate in redox and contribute to capacity fade



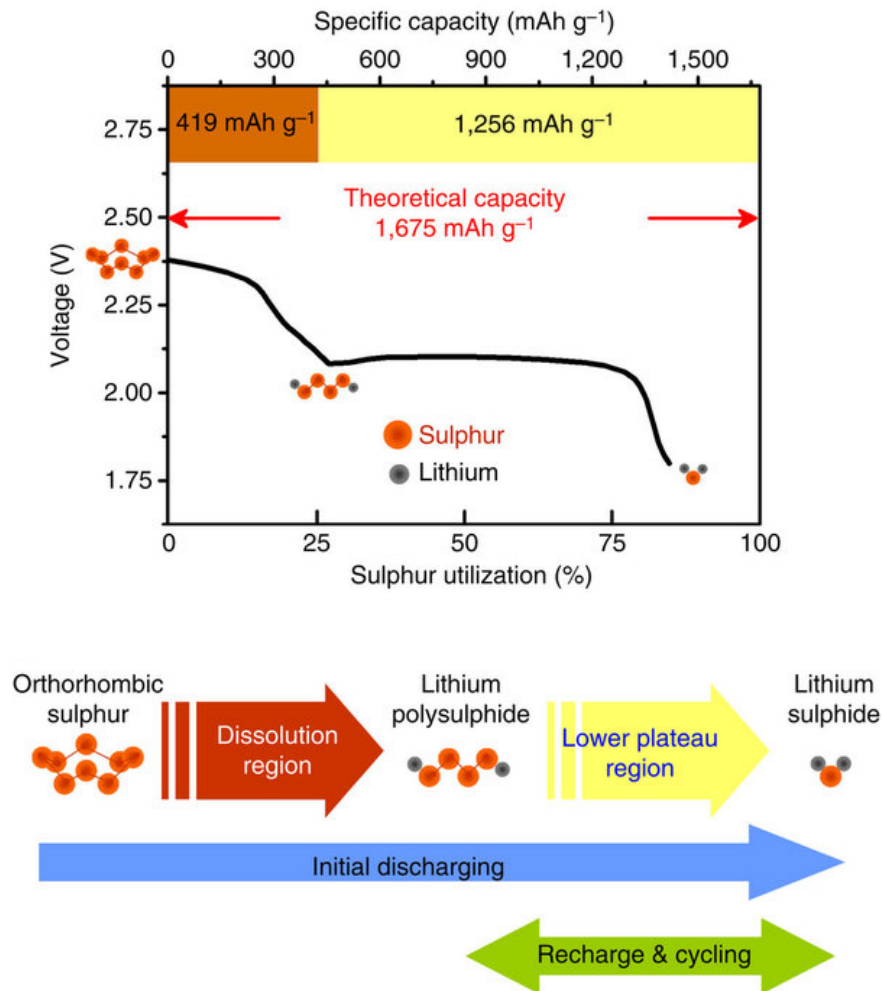
$\text{S}_2^{2-}$  near the cathode can further reduce to create  $\text{Li}_2\text{S}$ , adding to the capacity

Long-chain polysulfides shuttle to the anode and become reduced, creating short-chain polysulfides

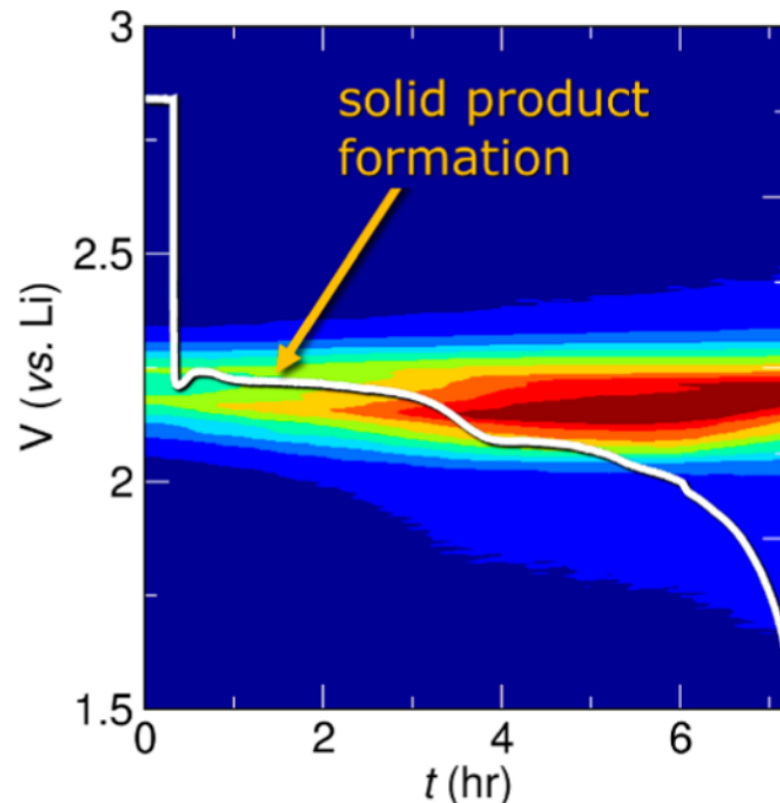
Short-chain polysulfides shuttle back to the cathode and form long-chain polysulfides again, leading to loss of active material (capacity fade)

# Debate of proposed discharge pathways

Series of solution reactions to form  $\text{Li}_2\text{S}$



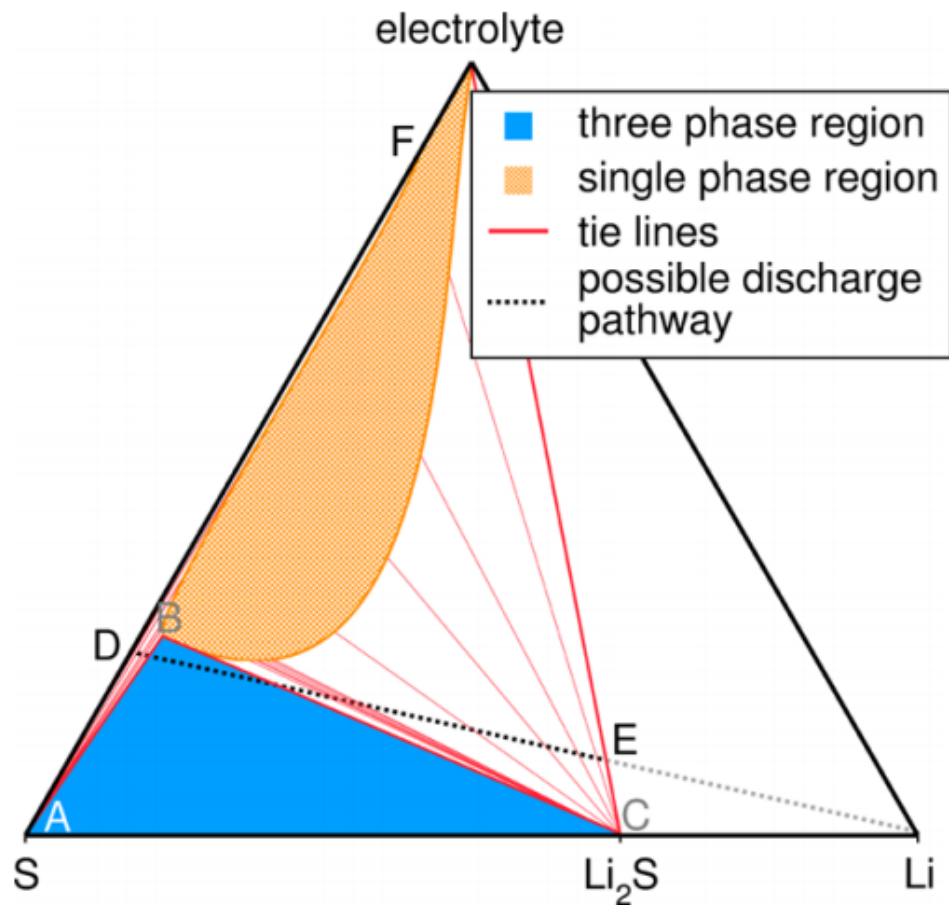
$\text{Li}_2\text{S}$  forms from the beginning



See, K. A.; Leskes, M.; Griffin, J. M.; Britto, S.; Matthews, P. D.; Emly, A.; Van der Ven, A.; Wright, D.S.; Morris, A.J.; Grey, C.P.; Seshadri, R. *JACS*, 2014, 136(46), 16368-16377.

Su, Y. S., Fu, Y., Cochell, T., Manthiram, A. A strategic approach to recharging lithium-sulphur batteries for long cycle life. *Nature communications*, 2013, 4(2985).

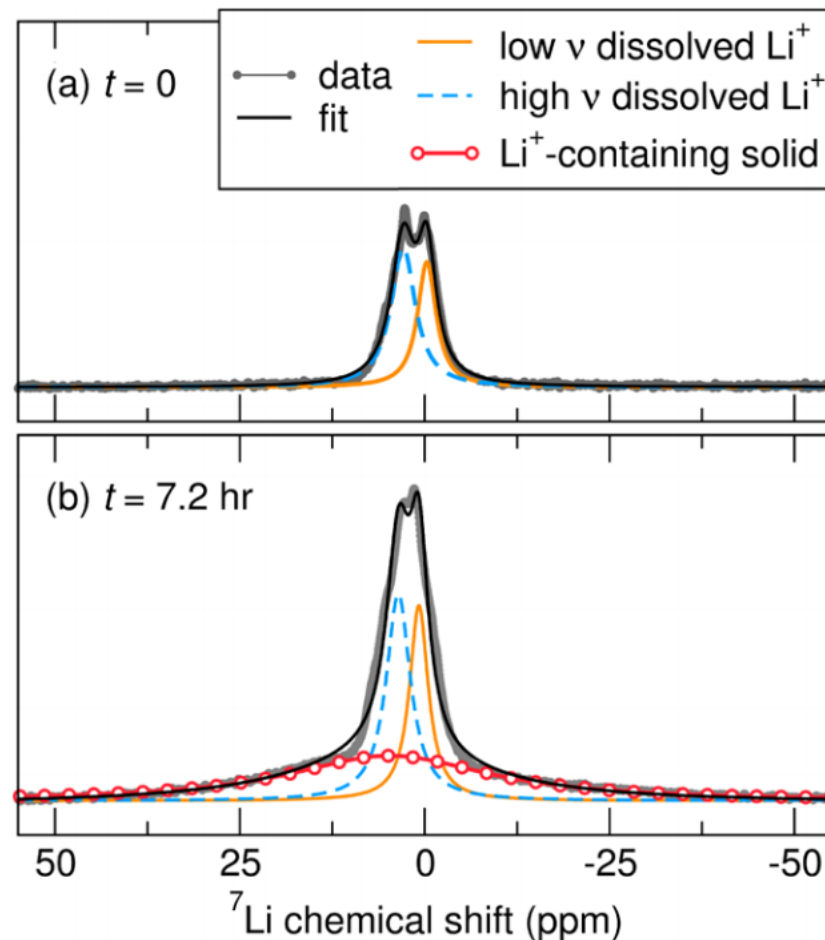
# Ab Initio Structure Search and in situ $^7\text{Li}$ NMR Studies of Discharge Products in the Li–S Battery System



See, K. A.; Leskes, M.; Griffin, J. M.; Britto, S.; Matthews, P. D.; Emly, A.; Van der Ven, A.; Wright, D.S.; Morris, A.J.; Grey, C.P.; Seshadri, R. *JACS*, 2014, *136*(46), 16368-16377.



# Benefit of in situ $^7\text{Li}$ NMR is the ability to characterize solids without crystallinity and dissolved species



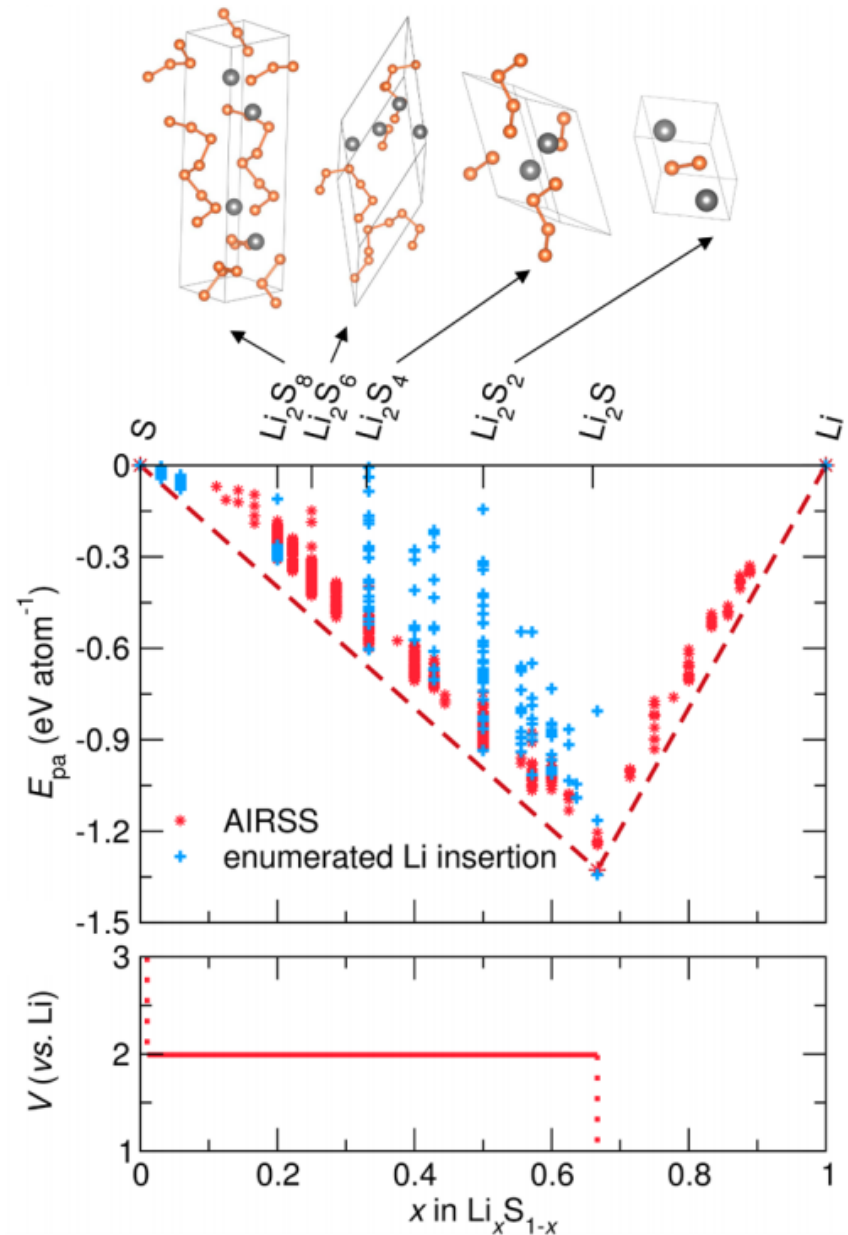
Monitor formation of  $\text{Li}_2\text{S}$  (red, broad peak)  
Distinguish between solid and dissolved products  
Low X-ray scattering powers of Li and S

# DFT calculations of the convex hull show $\text{Li}_2\text{S}$ is the only favored solid-state phase in this system

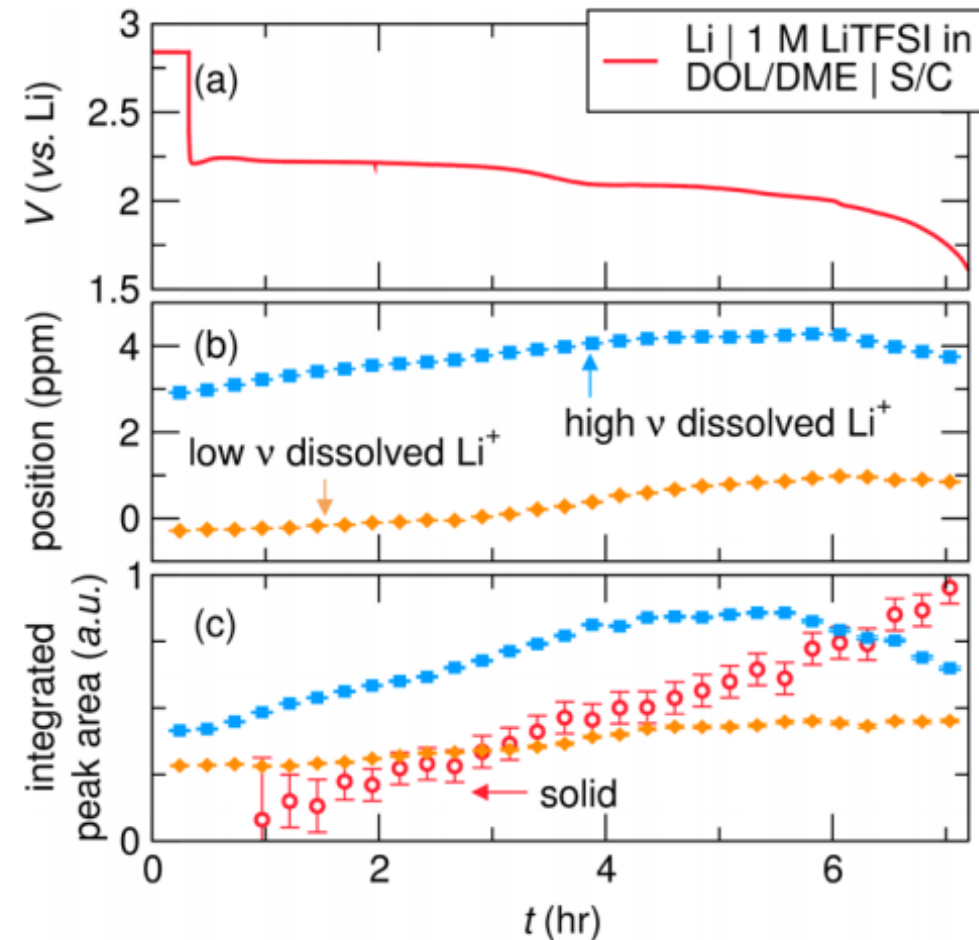
Lowest energy solid-state structures for  $\text{Li}_2\text{S}_8$ ,  $\text{Li}_2\text{S}_6$ ,  $\text{Li}_2\text{S}_4$ , and  $\text{Li}_2\text{S}_2$

Formation energies of possible stoichiometries of Li and S: all lie above the convex hull except Li, S, and  $\text{Li}_2\text{S}$

Discharge of an all-solid state battery would have one plateau corresponding to direct conversion to  $\text{Li}_2\text{S}$



# $^7\text{Li}$ NMR shows formation of $\text{Li}_2\text{S}$ forms from the beginning

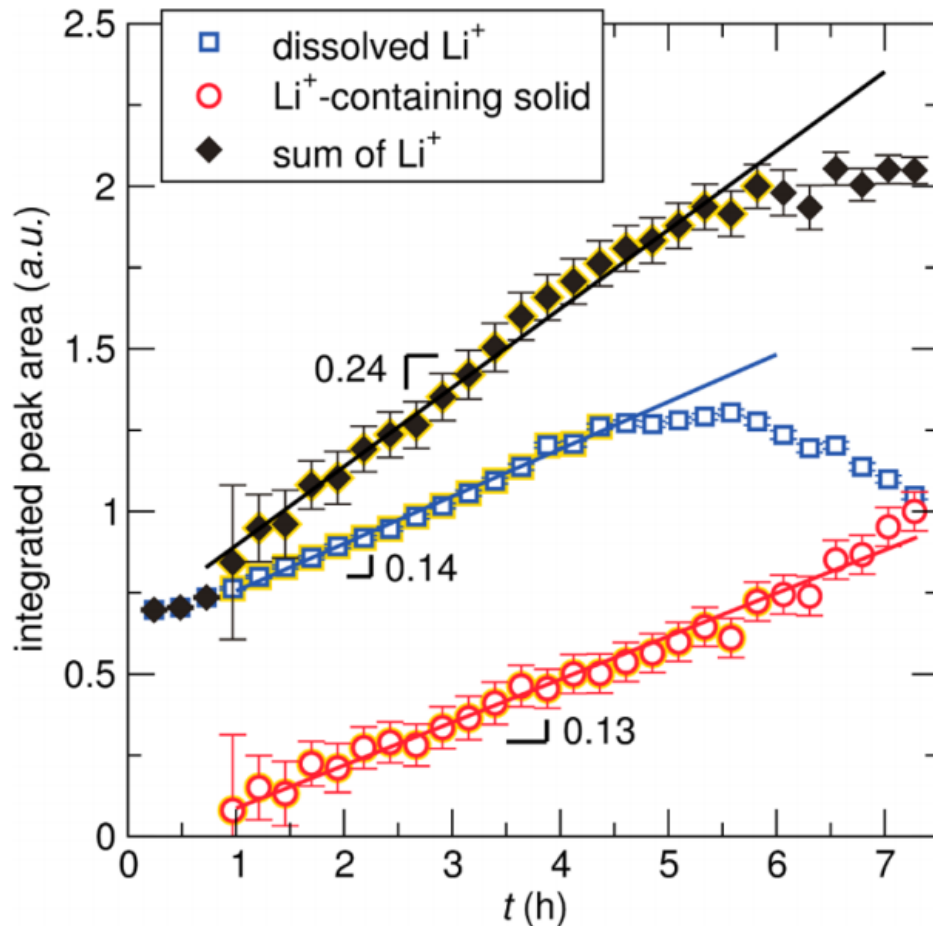


High frequency  $\text{Li}^+$  = closer to cathode  
High concentrations of  $\text{Li}^+$   
cause a positive shift in  
resonance

Low frequency  $\text{Li}^+$  = closer to anode  
Dissolution of polysulfides is  
relatively localized to the  
electrolyte near the cathode

Solid forms from beginning of discharge  
cycle

# $^7\text{Li}$ NMR confirms formation of $\text{Li}_2\text{S}$ is from electrochemical processes

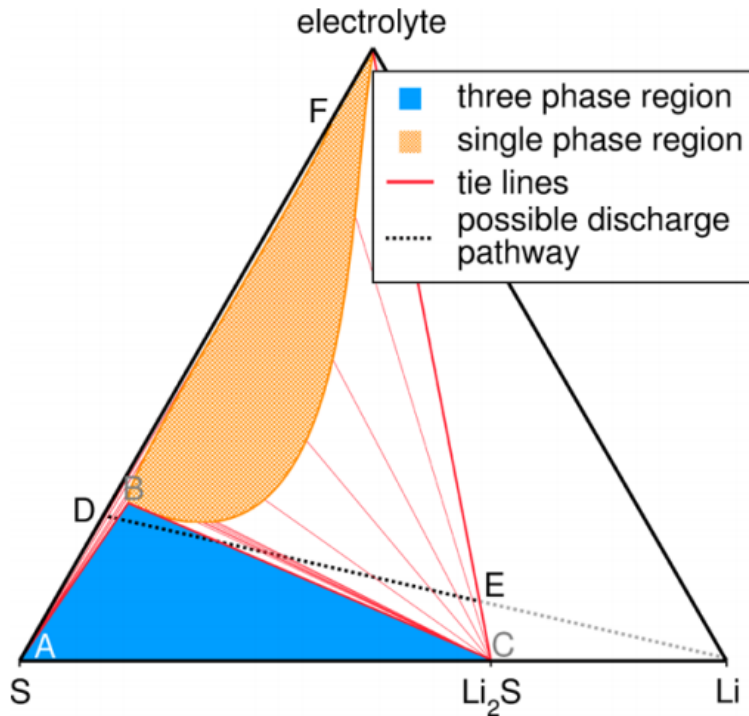


Integrated area of solid increases linearly with discharge

Suggests  $\text{Li}^+$ -containing solid is formed via electrochemical processes

For 1  $e^-$  pulled from the anode to produce 1  $\text{Li}^+$  ion, current and rate of formation of  $\text{Li}^+$  should be equal

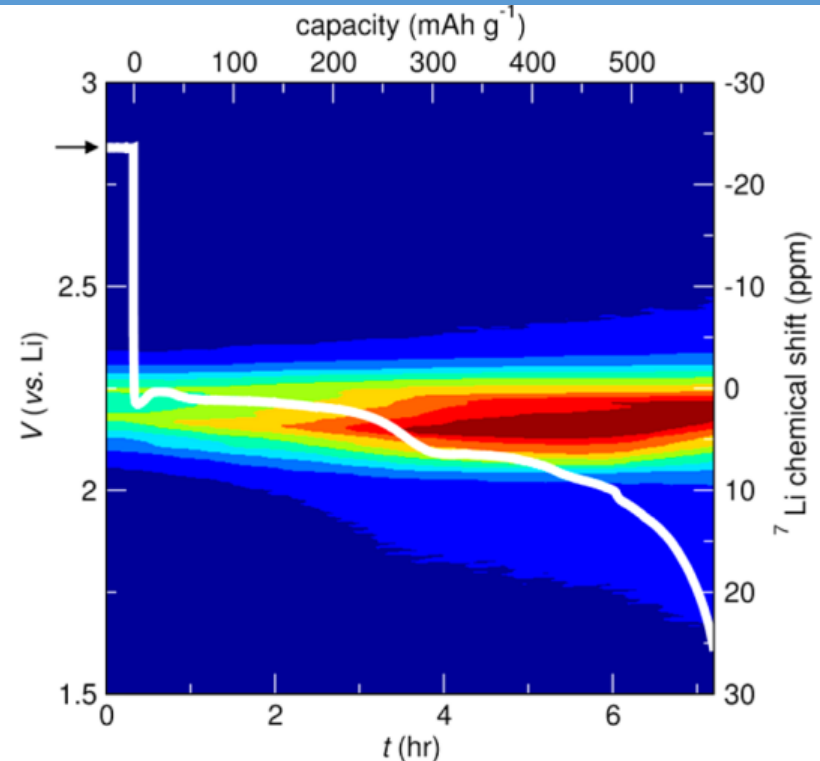
# Discharge mechanism best explained by ternary diagram



Single phase region: Li<sup>+</sup> and polysulfide species in solution

Two phase region: Li<sub>2</sub>S and electrolyte solvent

Three phase region: S (point A), dissolved polysulfides (point B), and Li<sub>2</sub>S (point C)



First plateau: discharge pathway passes through three phase region (form Li<sub>2</sub>S)

Drop in voltage: passage through set of dense tie lines between Li<sub>2</sub>S and stable dissolved polysulfide

Second plateau: S<sub>x</sub><sup>2-</sup> reduction from solution (incomplete) through two phase region