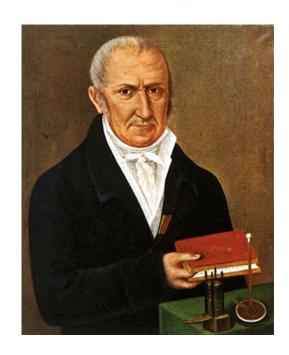
# A (Very Brief) History of The Battery





Alessandro Volta's battery (circa 1800 A.D.): copper and zinc separated by cardboard soaked in brine

However, jars have been discovered outside Baghdad dating to **200 B.C.**: iron rod, encased in copper, and soaked in vinegar or wine (0.78 V)

### **How Batteries Affect Your Life**







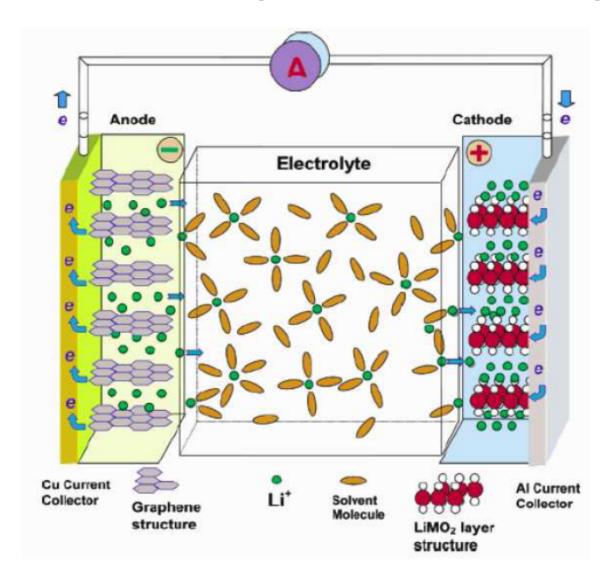




## The Main Components of a Typical Li-ion Battery

#### Components

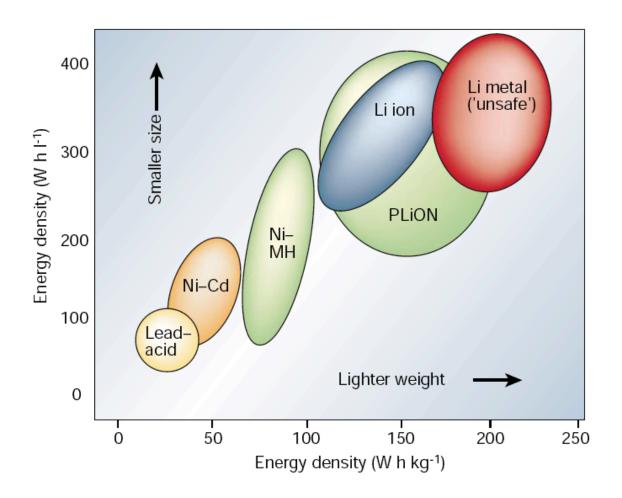
- Anode (Carbon)
- Cathode (LiCoO<sub>2</sub>)
- Electrolyte (EC/ DMC with LiPF<sub>6</sub>)
- Separator (Polypropylene)



# Topics to be discussed

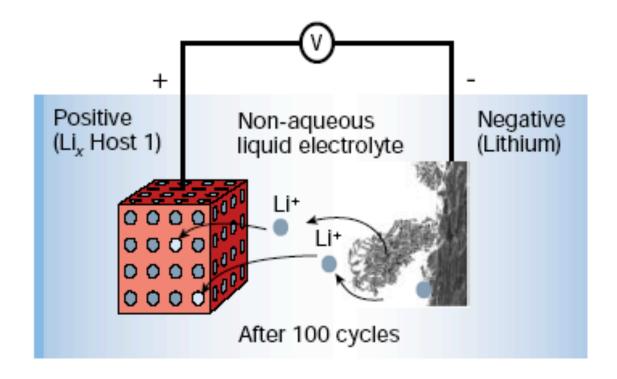
- Stability of Li (dendritic growth)
- Volume expansion challenges in electrode materials (anodes)
- Redox reactions in batteries
  - Energy level diagrams, which determine the electrochemical window of a battery
  - Solid-electrolyte-interfaces
- The importance of diffusion rates

### **Rich Battery Chemistry**



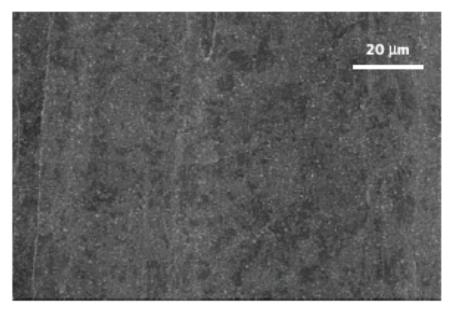
Advantages to Li: light, large electrochemical window

## **Intercalation Chemistry: Anodes**

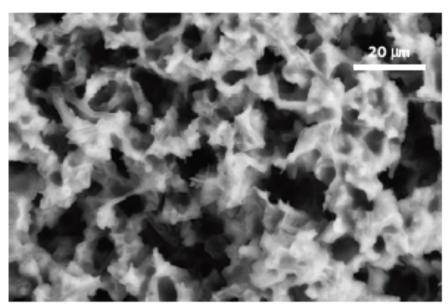


Dendritic growth of Li onto Li metal anodes

### Li-Dendrite Growth: Cause of Safety Failures



bare Li



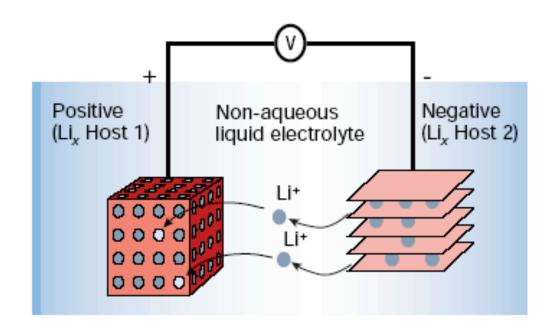
Li surface after charging

- Li dendrites puncture the separator, causing a short
- liquid electrolyte is flammable
- LiCoO<sub>2</sub> cathodes decompose to generate O<sub>2</sub>



Grey and Jerschow, Nature Materials 2012, 11, 311

## One solution: Graphite



Current commercial batteries have graphite anodes trade-off is lower energy density

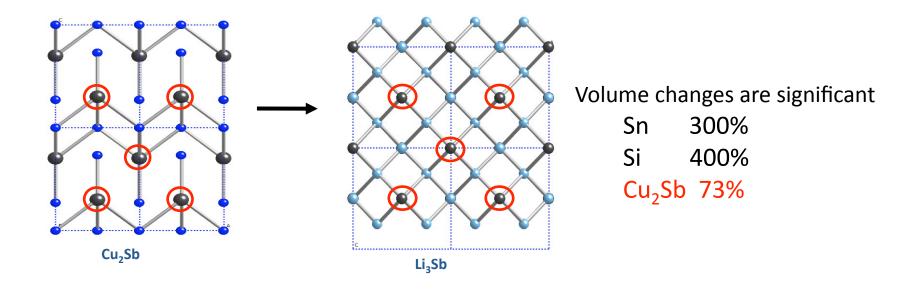
### Cu<sub>2</sub>Sb

Graphite based materials are the current standard

Theoretical capacity of 372 mAh g<sup>-1</sup> (818 mAh ml<sup>-1</sup>) (LiC<sub>6</sub>)

Cu<sub>2</sub>Sb improves cyclic stability

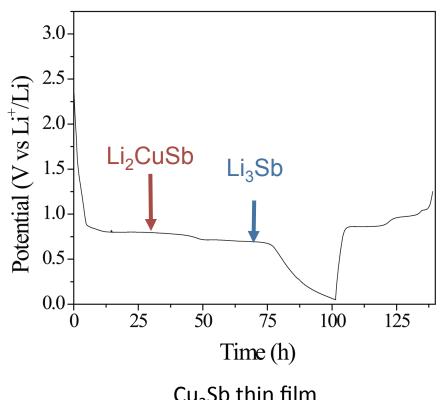
Experimental value of 290 mAh g<sup>-1</sup> (1914 mAh ml<sup>-1</sup>)



Strong structural relationship between parent and child compound No danger of plating Li metal

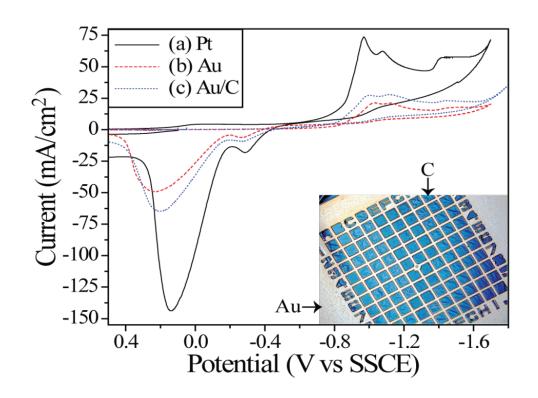
# Lithiation of Cu<sub>2</sub>Sb

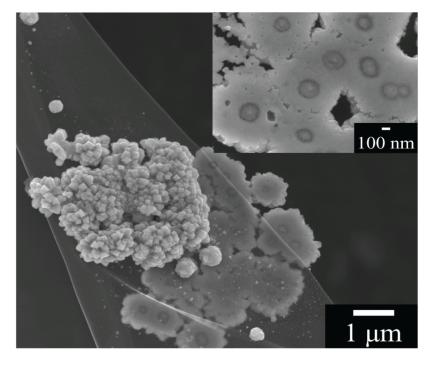
- Lithiation of Cu<sub>2</sub>Sb has two main characteristic plateaus:
  - 1<sup>st</sup> at ~0.8 V vs. Li/Li<sup>+</sup>  $Cu_2Sb \longrightarrow Li_2CuSb$
  - 2<sup>nd</sup> at ~0.6 V vs. Li/Li<sup>+</sup>
     Li<sub>2</sub>CuSb → Li<sub>3</sub>Sb
- The charge before 0.8 V is due to SEI formation and formation of Li<sub>2</sub>O (if oxide is present)



 $Cu_2Sb$  thin film C/100 1M LiClO<sub>4</sub> (1:1:1 by vol.) EC:DEC:DMC

# **Depositing onto TEM Grids**

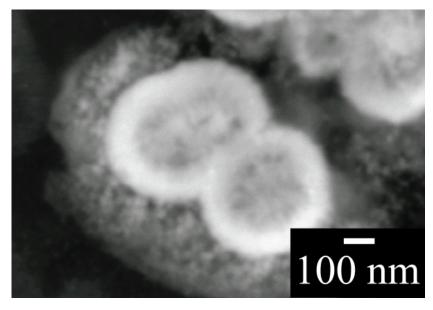




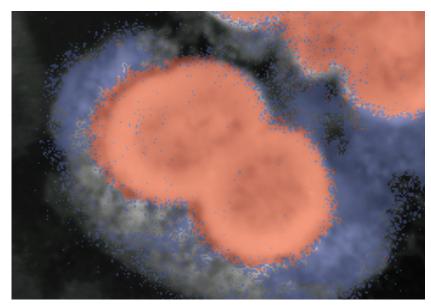
J. Mosby, D. C. Johnson, A. L. Prieto, J. Electrochem. Soc. 2010, 157, E99

### First Charge Plateau

The 1<sup>st</sup> step in the lithiation process occurs at 0.8 V vs Li/Li<sup>+</sup>
Cu<sub>2</sub>Sb + 2e<sup>-</sup> +2Li<sup>+</sup> → Li<sub>2</sub>CuSb + Cu
~0.4 mole to 1 mole of Cu per mole of Cu<sub>2</sub>Sb is extruded



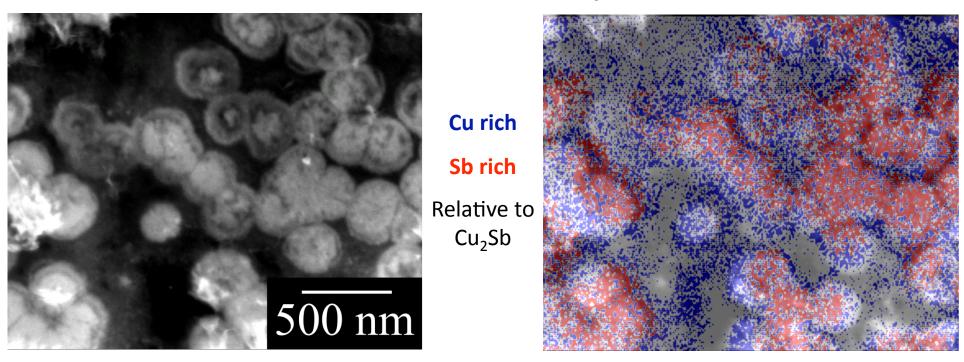
Cu rich
Sb rich
Relative to  $Cu_2Sb$ 



The extruded Cu particles surround the nucleation sites

### **Second Charge Plateau**

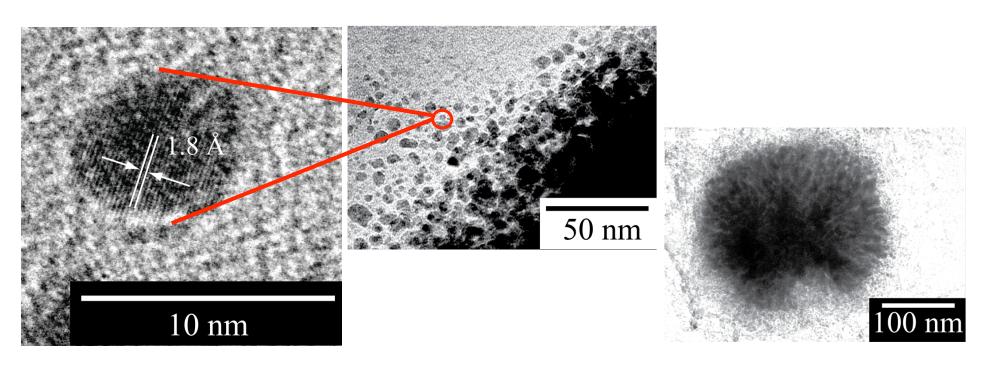
The 2<sup>nd</sup> step in the lithiation process occurs at 0.6 V vs Li/Li<sup>+</sup> Li<sub>2</sub>CuSb + 1e<sup>-</sup> + Li<sup>+</sup> → Li<sub>3</sub>Sb + Cu



Extruded copper is further from nucleation site

#### **Extruded Cu Particles**

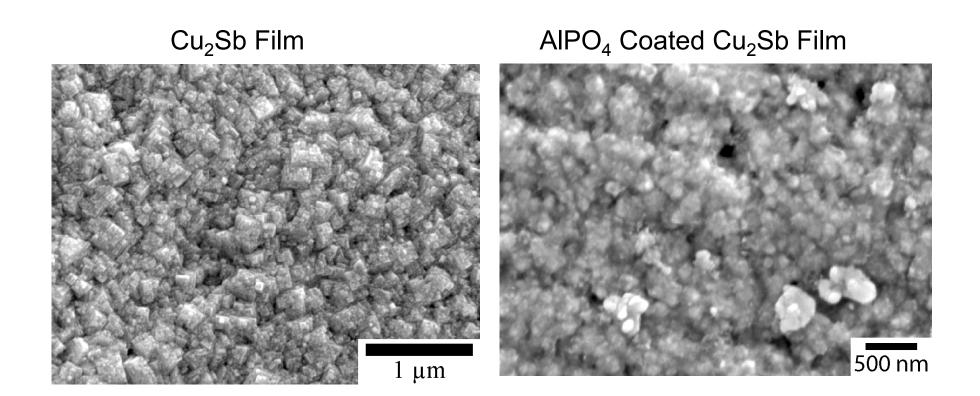
The extruded Cu is in the form of small < 10 nm particles



By 0.6 V the Cu particles have traveled from the core and lattice fringes that match Cu(200) are present

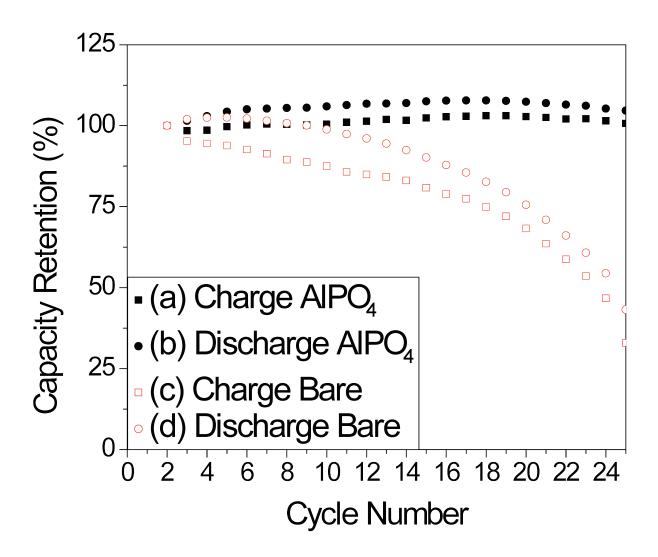
For examples of other systems that extrude metal particles see E. Takeuchi (Ag)

# Using a Physical Barrier to Keep the Cu in Contact



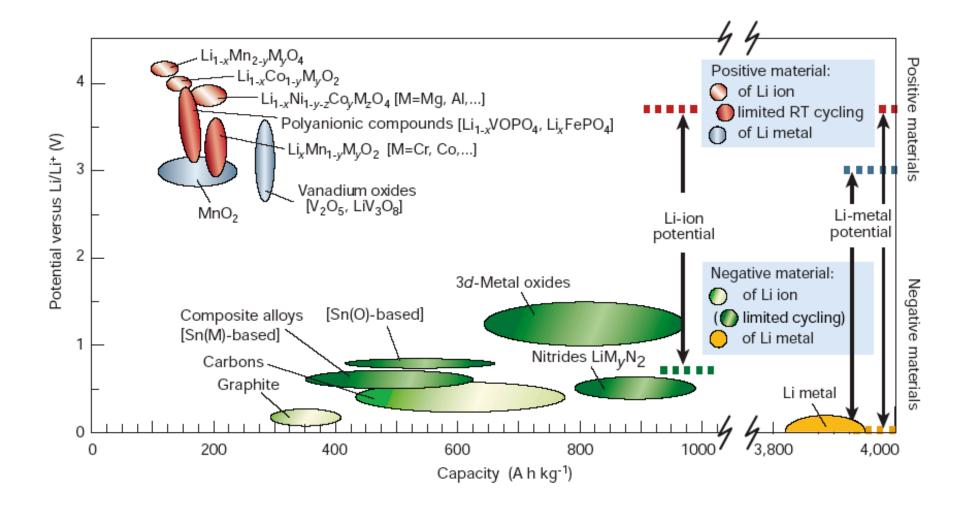
AlPO<sub>4</sub> can be coated onto the Cu<sub>2</sub>Sb surface 'electrochemically'

### **Enhanced Capacity Retention**



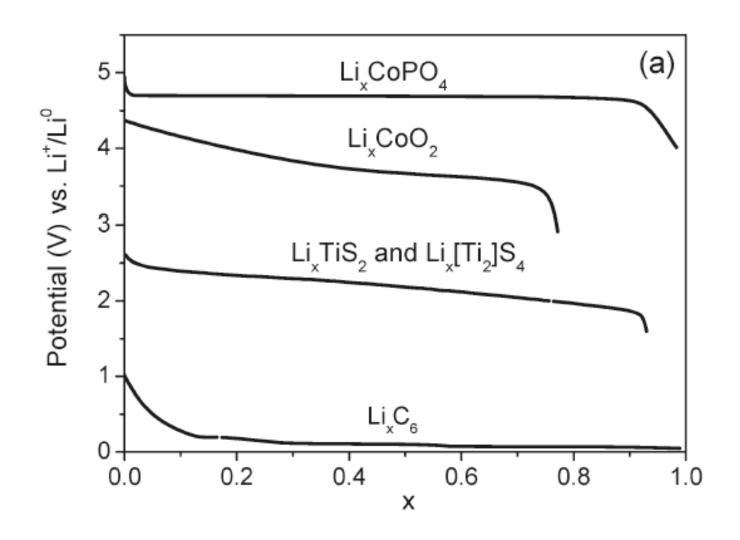
AlPO<sub>4</sub> coated Cu<sub>2</sub>Sb films have a higher capacity retention

#### **Different Chemistries**

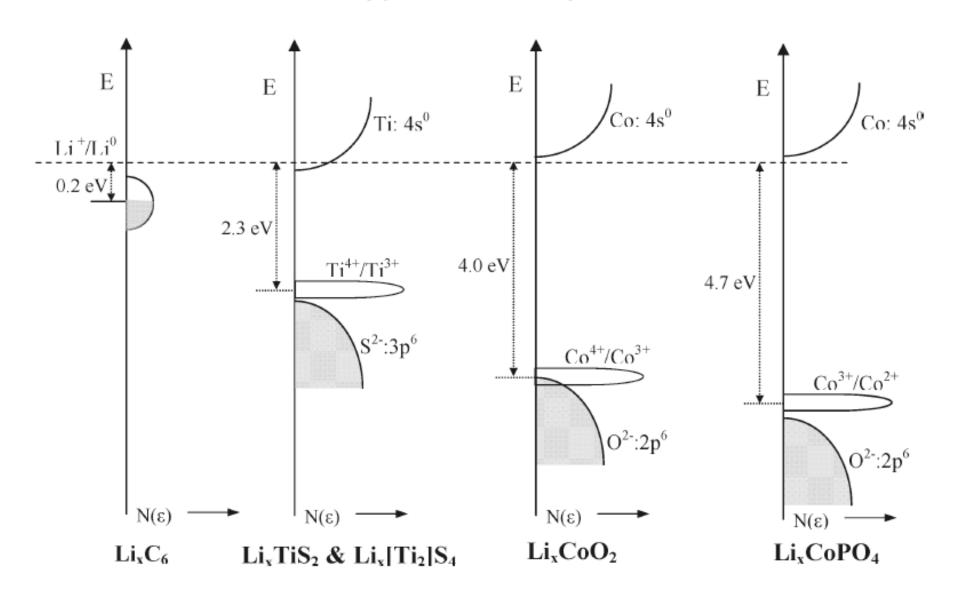


There is still a significant interest in solving the Li-dendrite problem.

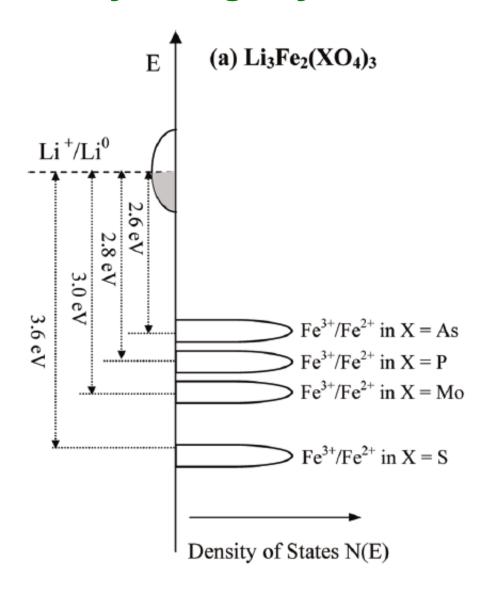
# Voltage profiles of common cathodes



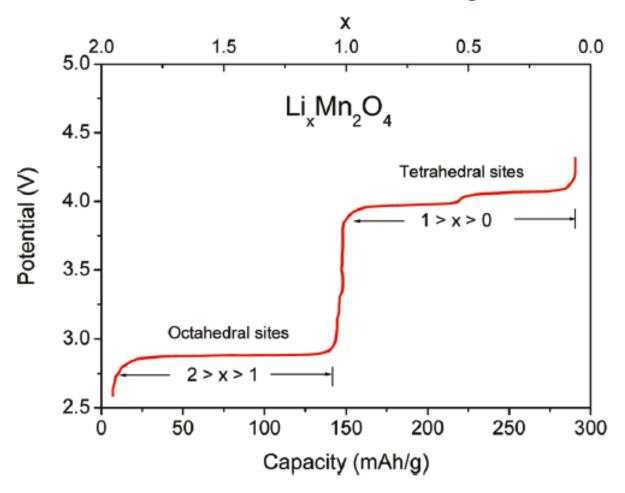
# **Energy level diagrams**



### **Tunability Using Crystal Chemistry**

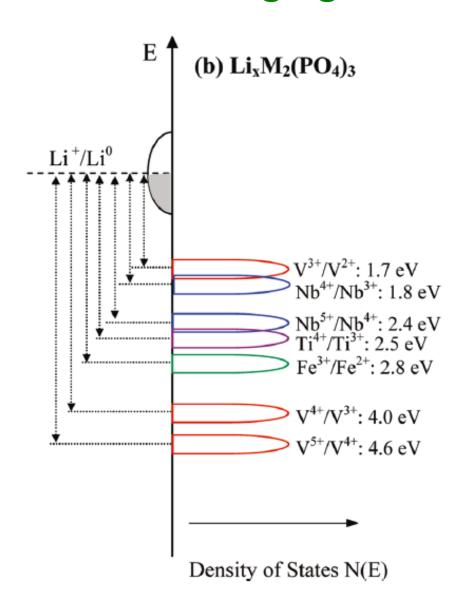


### **Effect of Geometry**



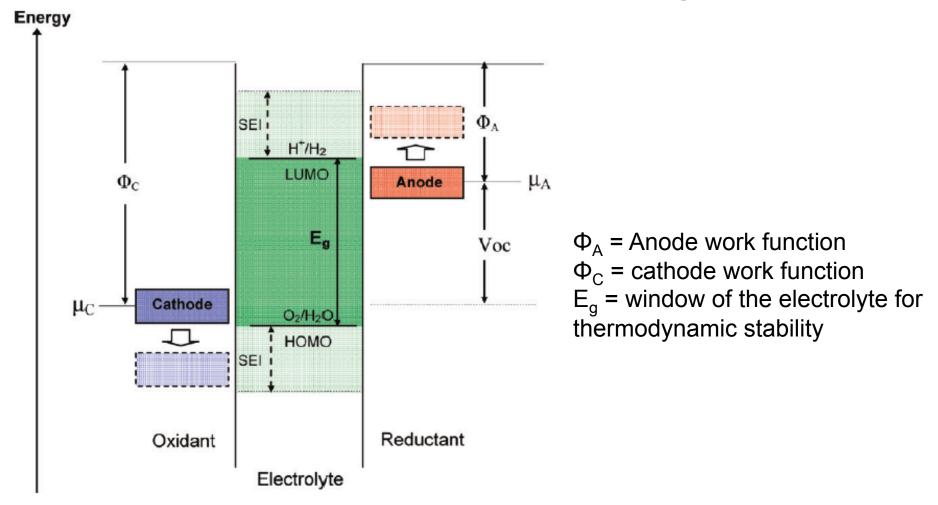
But if you take all of the Li+ out, the structure collapses and it isn't reversible.

### **Changing the transition metal**



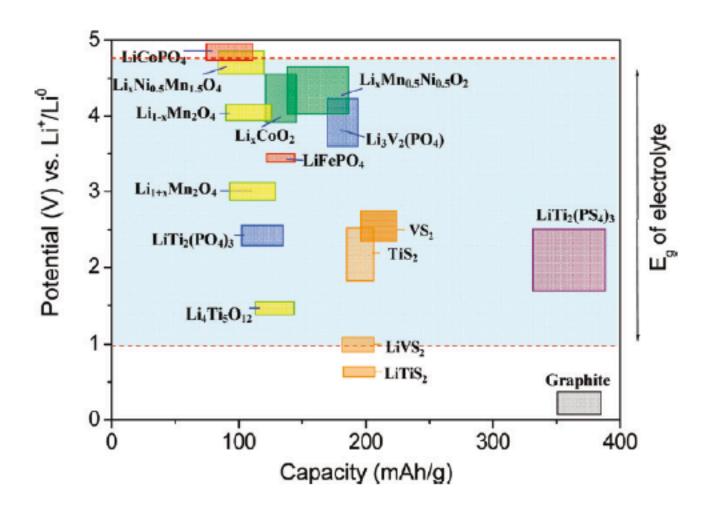
The challenge is that when you make the voltages large, you have to worry about unwanted reactions with the electrolyte

### **Liquid Electrolyte Stability**



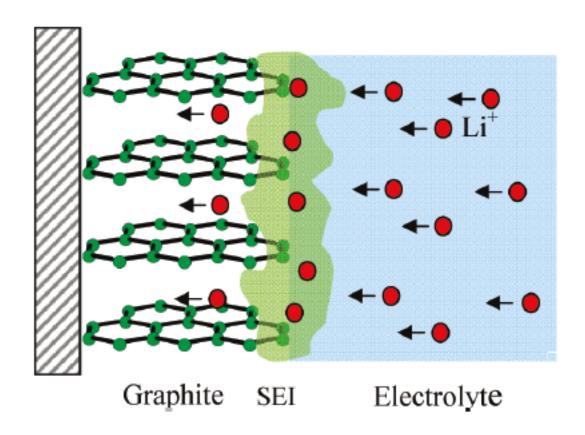
The kintetic stability of a battery is due to the formation of a stable **s**olid **e**lectrolyte **i**nterface (SEI)

### Voltage window of the electrolyte



Voltage versus capacity relative to the window of 1 M LiPF<sub>6</sub> in EC/DEC (1:1)

#### **Generic Cartoon of the SEI**



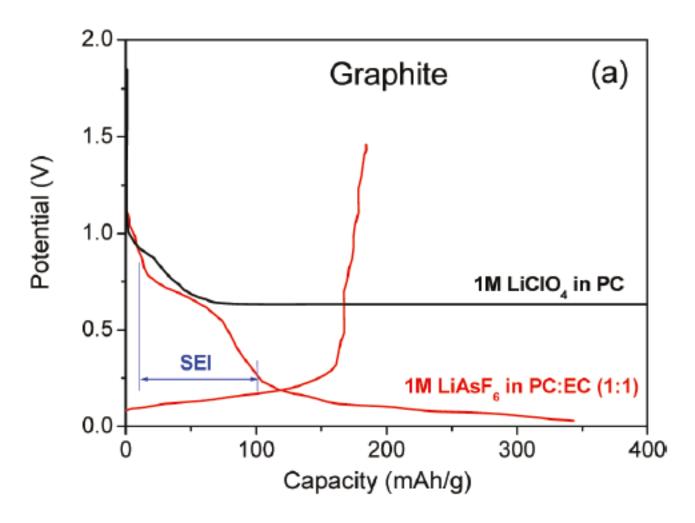
The SEI is still poorly characterized, although people have developed ways to control its growth with the addition of various additives

# **Summary of electrolytes**

Table 1. Nonaqueous Electrolytes for Li-Ion Batteries

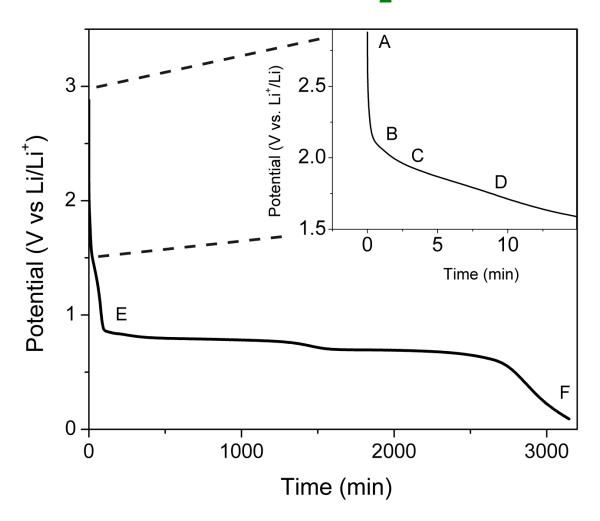
Electrolytes	Example of classical electrolytes	Ionic conductivity $(\times 10^{-3} \text{ s/cm})$ at room temp	Electrochemical window (V) vs Li <sup>+</sup> /Li <sup>0</sup>		
			Reduction	Oxidation	Remark
Liquid organic	1M LiPF <sub>6</sub> in EC:DEC (1:1) 1M LiPF <sub>6</sub> in EC:DMC (1:1)	$7^3$ $10^3$	1.3 <sup>7</sup> 1.3 <sup>7</sup>	4.5 <sup>6</sup> > 5.0 <sup>3</sup>	Flammable
Ionic liquids	1M LiTFSI in EMI-TFSI 1M LiBF <sub>4</sub> in EMI-BF <sub>4</sub>	$2.0^{15} \\ 8.0^{15}$	$\begin{array}{c} 1.0^{15} \\ 0.9^{16} \end{array}$	5.3 <sup>15</sup> 5.3 <sup>16</sup>	Non-flammable
Polymer	LiTFSI-P(EO/MEEGE) LiClO <sub>4</sub> –PEO <sub>8</sub> + 10 wt % TiO <sub>2</sub>	$0.1^{24} \\ 0.02^{26}$	$< 0.0^{24} < 0.0^{26}$	$4.7^{24} \\ 5.0^{26}$	Flammable
Inorganic solid	$\text{Li}_{4-x}\text{Ge}_{1-x}\text{P}_{x}\text{S}_{4} (x = 0.75)$ $0.05\text{Li}_{4}\text{SiO}_{4} + 0.57\text{Li}_{2}\text{S} + 0.38\text{SiS}_{2}$	$2.2^{28} \\ 1.0^{30}$	$< 0.0^{28} < 0.0^{30}$	$> 5.0^{28}$ $> 8.0^{30}$	Non-flammable
Inorganic liquid	$LiAlCl_4 + SO_2$	$70^{20}$	-	$4.4^{20}$	Non-flammable
Liquid organic + Polymer	$0.04 \text{LiPF}_6 + 0.2 \text{EC} + 0.62 \text{DMC} + 0.14 \text{PAN}$	$4.2^{38}$	-	$4.4^{38}$	Flammable
	$LiClO_4 + EC + PC + PVdF$	$3.0^{39}$	-	$5.0^{39}$	
Ionic liquid + Polymer	$\begin{array}{l} 1M\ LiTFSI + P_{13}TFSI + \\ PVdF\text{-}HFP \end{array}$	0.18 <sup>43</sup>	< 0.0 <sup>43</sup>	5.8 <sup>43</sup>	Less flammable
Ionic liquid + Polymer + Liquid organic	56 wt % LiTFSI-Py <sub>24</sub> TFSI + 30 wt % PVdF-HFP + 14 wt % EC/PC	0.8144	1.5 <sup>44</sup>	4.2 44	Less flammable
Polymer + Inorganic solid	2 vol % LiClO <sub>4</sub> -TEC-19 + 98 vol% 95 (0.6Li <sub>2</sub> S + 0.4Li <sub>2</sub> S) + 5Li <sub>4</sub> SiO <sub>4</sub>	0.0346	< 0.0 <sup>46</sup>	>4.5 <sup>46</sup>	Non-flammable
Ionic liquid + Liquid organic <sup>19</sup>		-	-	-	Non-flammable

# **SEI Formation on the First Cycle**



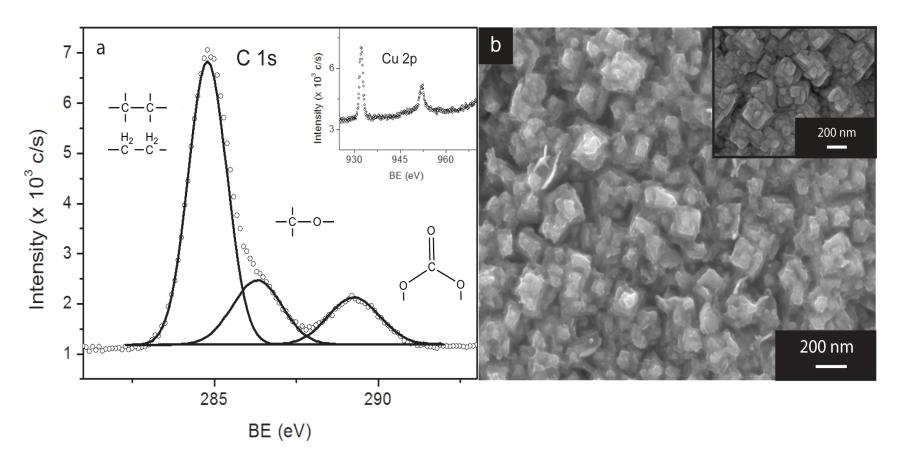
Industrial batteries are cycled several times before they are sold: forming process

### Back to Cu<sub>2</sub>Sb



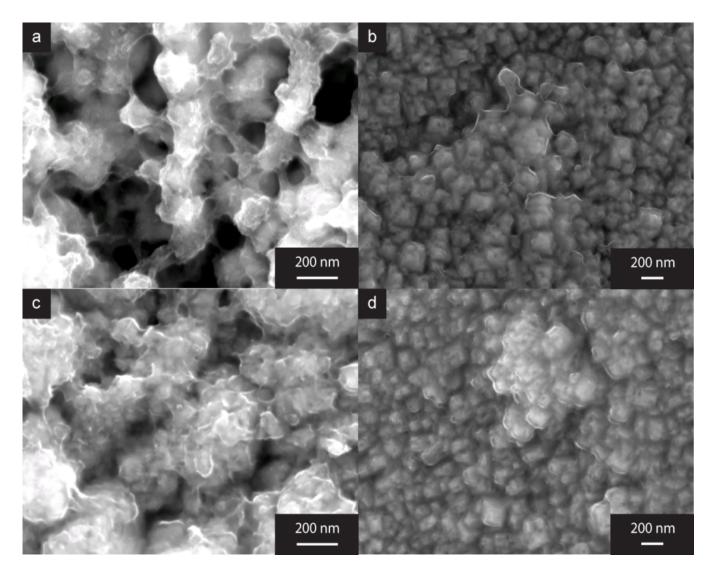
Cu<sub>2</sub>Sb electrochemical half-cell galvanostatically charged at 10 μA cm<sup>-2</sup> in a 1M LiClO<sub>4</sub> EC/DEC/DMC (1:1:1) electrolyte solution

# Decomposition of Liquid Electrolyte on Cu<sub>2</sub>Sb



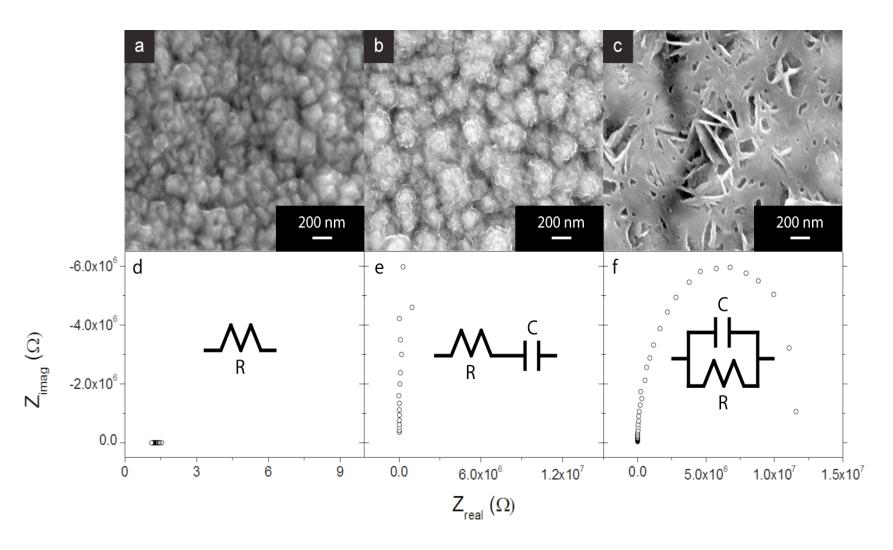
(a) XPS HRES C 1s and Cu 2p (inset) spectra of a Cu<sub>2</sub>Sb electrode charged to 1.8 V vs Li/Li<sup>+</sup> (b) SEM micrographs of the SEI layer formed on the surface of the Cu<sub>2</sub>Sb. The cubic morphology of the bare electrode (inset) is still present under the SEI layer

# Potential Range is Important in Morphology



SEM micrographs of an SEI grown in the MPR for (a) 5 cycles and (c) 20 cycles. The SEI formed in the HPR for (b) 5 cycles and (d) 50 cycles.

### Potential Range is Important in Performance



High potentials

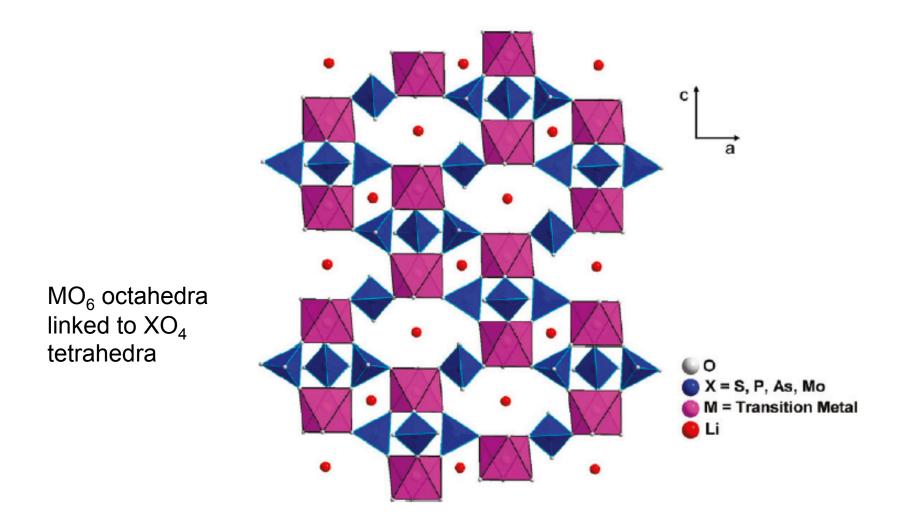
intermediate potentials

low potentials

### Take Home Messages Thus Far

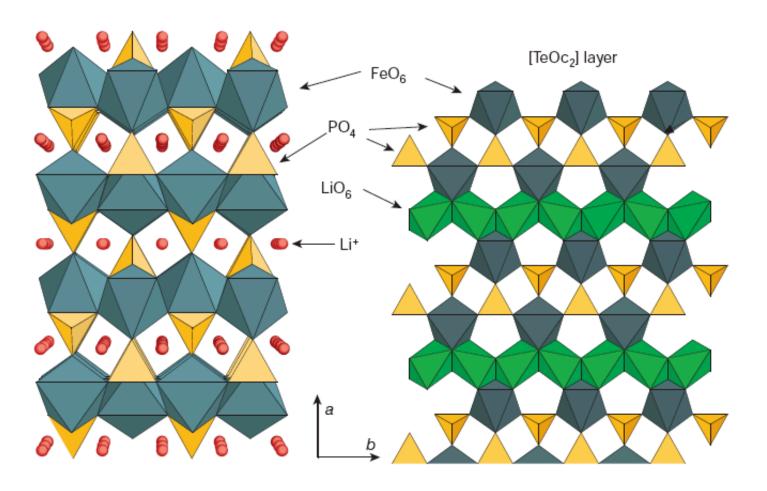
- Stability of Li (dendritic growth)
  - This hasn't been solved on Li metal anodes, and is a significant safety hazard
  - Alternative anode materials can be safer, but you lose on energy density and voltage
- Volume expansion challenges in electrode materials (anodes)
  - Mechanical pulverization is a problem in all electrode materials, but especially in dense intermetallics
  - structural similarities between the parent and child compounds can be used to reduce the overall volume changes
- Solid-electrolyte-interfaces
  - Big voltage windows lead to redox reactions with the electrolyte
  - Interfaces are critical in batteries!
- The importance of diffusion rates

#### Framework Structures: NASICON



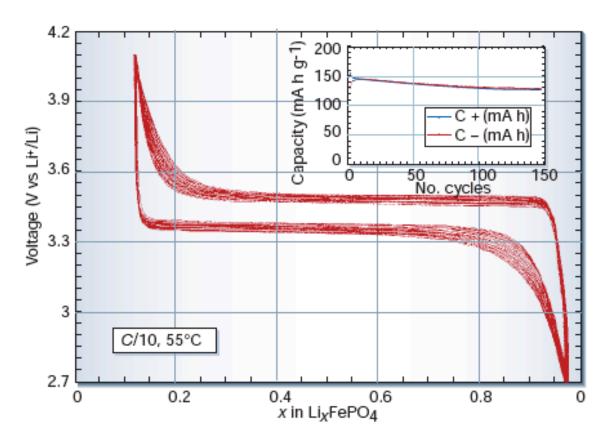
The diffusion rates of Li+ ions into solid state structures is critical for *how fast* your battery will charge and discharge.

# A123's Cathode Material: LiFePO<sub>4</sub>



Li+ ions in 1D channels.

### **Charging Behavior: Must be Mixed With Carbon**



C rate = capacity/time, or C/n denotes the rate at which a full charge or discharge takes in hours

Advantages: stable over a wide composition range

### A Current Example: Kang and Ceder

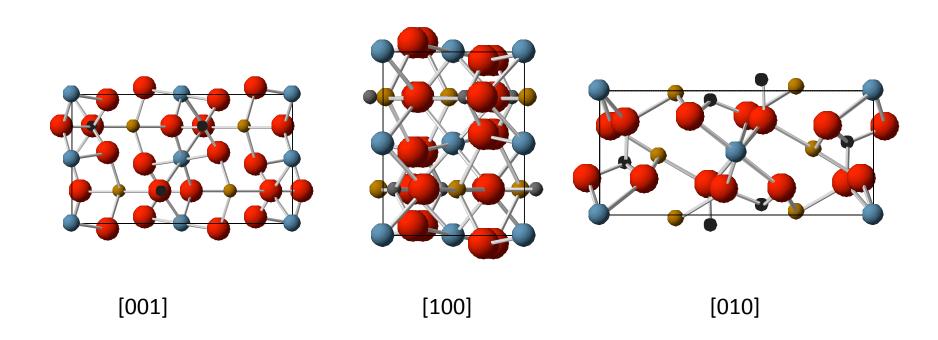
Battery materials for ultrafast charging and discharging

Nature 2009

The storage of electrical energy at high charge and discharge rate is an important technology in today's society, and can enable hybrid and plug-in hybrid electric vehicles and provide back-up for wind and solar energy. It is typically believed that in electrochemical systems very high power rates can only be achieved with supercapacitors, which trade high power for low energy density as they only store energy by surface adsorption reactions of charged species on an electrode material <sup>1-3</sup>. Here we show that batteries <sup>4,5</sup> which obtain high energy density by storing charge in the bulk of a material can also achieve ultrahigh discharge rates, comparable to those of supercapacitors. We realize this in LiFePO<sub>4</sub> (ref. 6), a material with high lithium bulk mobility <sup>7,8</sup>, by creating a fast ion-conducting surface phase through controlled off-stoichiometry. A rate capability equivalent to full battery discharge in 10–20 s can be achieved.

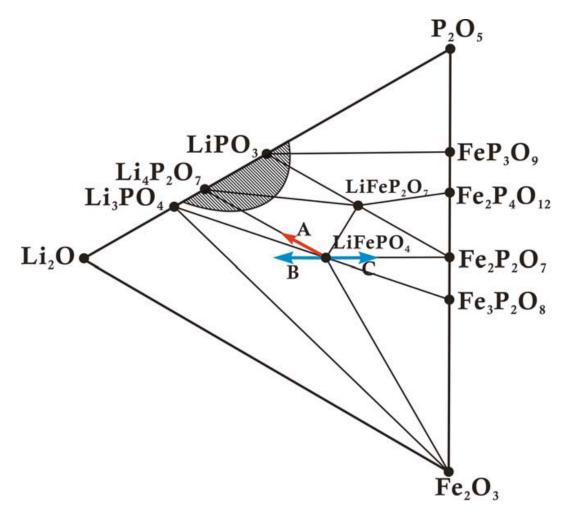
### Diffusion of Li<sup>+</sup> is Direction Dependent

Kang and Ceder explained that Li<sup>+</sup> ions move into the bulk of LiFePO<sub>4</sub> primarily through the **[010] direction**. The oxygen is shown in red, lithium is blue, phosphorous is grey, and iron is mustard yellow.



[010] has fairly open channels....ideal for cation diffusion

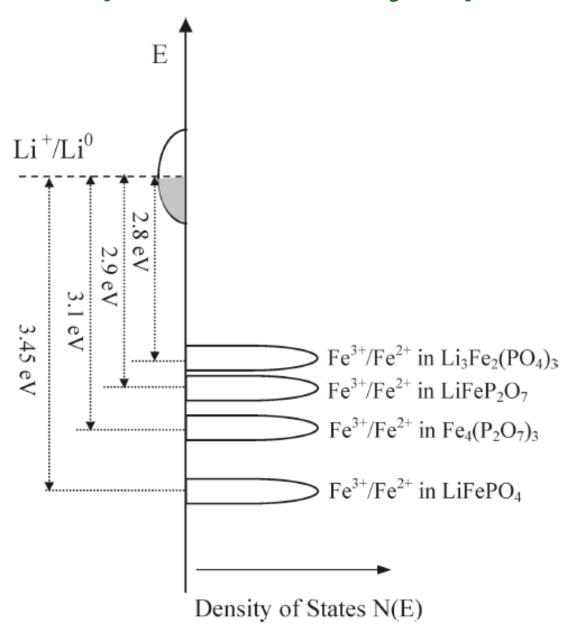
### **Ternary Phase Diagram**



A: poorly crystallized lithium phosphate (good Li conductors)

**B**: Li-excess or **C**: Li-deficiency leads to Fe-oxides or Fe-phosphates

# **Control of Composition is Really Important!**



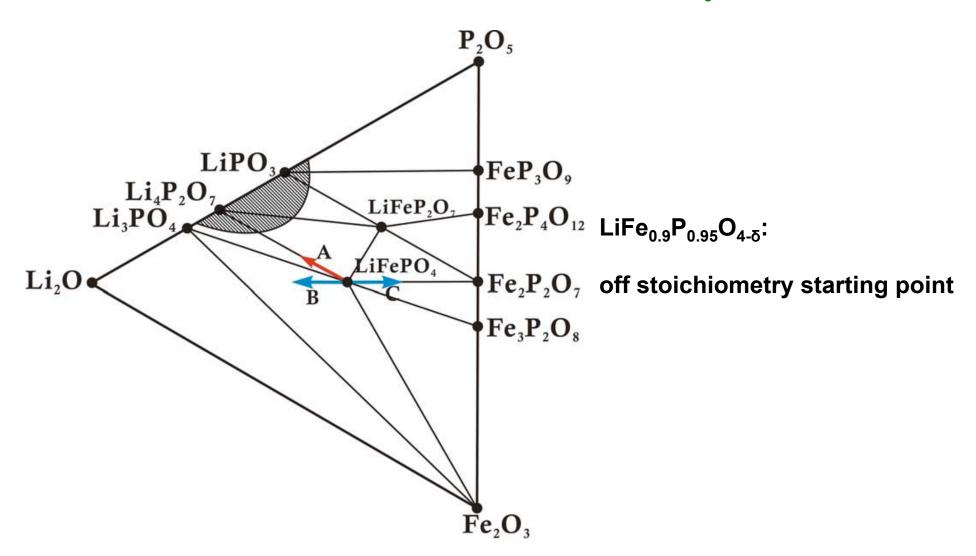
### Making A 'Coated' Battery Material

Our synthesis strategy has been to create an appropriate offstoichiometry in the starting materials so that the coating constituents phase-separate from LiFePO<sub>4</sub> as it forms during the heat treatment, thereby creating the active storage material and coating in a single process. Here we describe results with an iron:phosphorus deficiency ratio of 2:1 (for example LiFe<sub>1-2y</sub>P<sub>1-y</sub>O<sub>4- $\delta$ </sub>, y = 0.05), as indicated by arrow A in Supplementary Fig. 1. We note that the more common one-to-one iron:phosphorus deficiency (arrow B in Supplementary Fig. 1, equivalent to lithium excess<sup>22</sup>) creates a mixture of Li<sub>3</sub>PO<sub>4</sub> and iron oxides, which are not likely to conduct well under the synthesis conditions used to prepare LiFePO<sub>4</sub>.

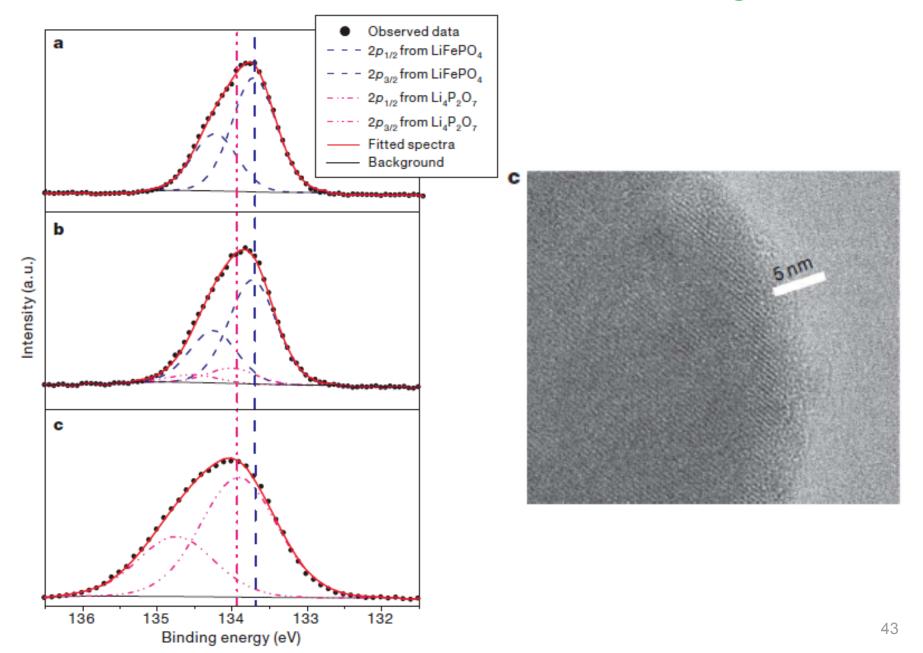
### **Synthesis**

LiFe<sub>0.9</sub>P<sub>0.95</sub>O<sub>4- $\delta$ </sub> was synthesized by ball-milling Li<sub>2</sub>CO<sub>3</sub>, FeC<sub>2</sub>O<sub>4</sub>·2H<sub>2</sub>O and NH<sub>4</sub>H<sub>2</sub>PO<sub>4</sub> in appropriate amounts, heating the mixture at 350 °C for 10 h and then heating at 600 °C for 10 h under argon. X-ray diffraction (Fig. 1a and Supplementary Fig. 2) shows that despite the off-stoichiometric starting mixture, stoichiometric LiFePO<sub>4</sub> forms with lattice parameters (a = 10.3134 Å, b = 6.002 Å and c = 4.691 Å) very similar to those reported in the literature<sup>23</sup>.

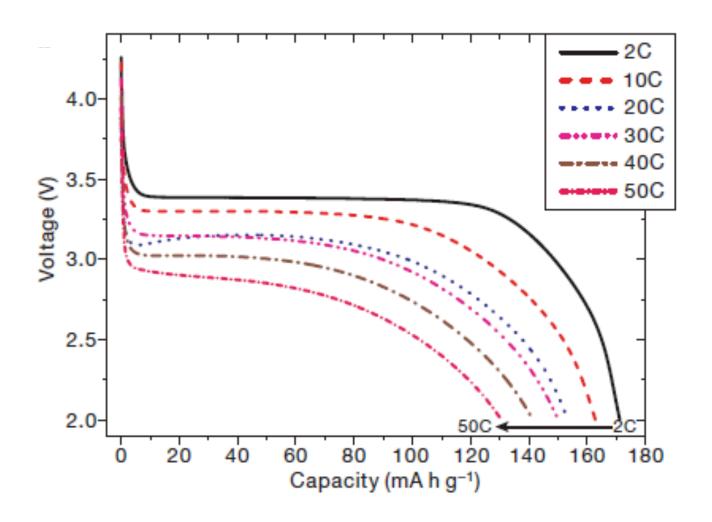
# Aiming for LiFePO<sub>4</sub> and Li<sub>x</sub>P<sub>y</sub>O<sub>1-δ</sub>



# **Data for Surface Coating**

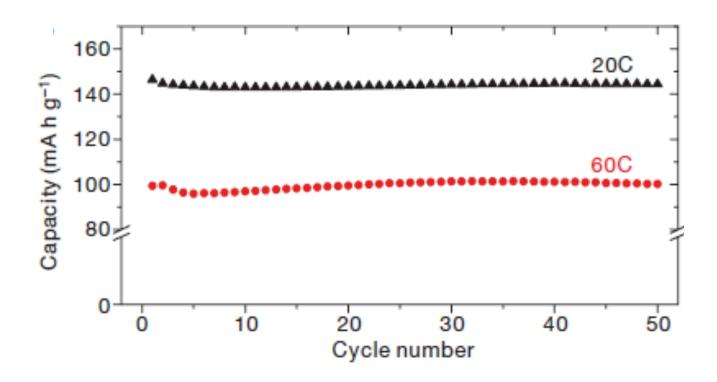


#### **Incredible Rate Performance**



The capacity retention is really impressive at high rates. 50C is completely discharged in 72s

# **Cycle Life at High Rates**



You still see a capacity loss at high rates, but it's among the best performing fast rate materials currently known.

competitive with supercapacitors

# **Final Take Home Messages**

- Stability of Li (dendritic growth)
  - This hasn't been solved on Li metal anodes, and is a significant safety hazard
  - Alternative anode materials can be safer, but you lose on energy density and voltage
- Volume expansion challenges in electrode materials (anodes)
  - Mechanical pulverization is a problem in all electrode materials, but especially in dense intermetallics
  - structural similarities between the parent and child compounds can be used to reduce the overall volume changes
- Solid-electrolyte-interfaces
  - Big voltage windows lead to redox reactions with the electrolyte
  - Interfaces are critical in batteries!
- The importance of diffusion rates
  - Can improve diffusion by nanostructuring (Prof. Bruce)
  - Structural features are critical

Questions? You can reach me at alprieto@lamar.colostate.edu