# Transparent Conductors Part II

Prof. Tom Mason Northwestern University Materials Science & Engineering

# Conclusions

- With the exception of ZnO, the best TCOs are high octahedral site density crystal structures
- They are n-type only
- The "basis" cations are In, Sn, Zn, Cd and Ga
- Their defect chemistries can be complicated (not just oxygen vacancies)
- Phase diagrams play an important role
- If you can make a comparable p-type TCO...

# An Introduction to TCOs

Transparency (> 80% through visible spectrum for typical 1 μm thin film)



Conductivity (> 10<sup>3</sup> S/cm)

- What About p-Type TCOs
- The Importance of Work Functions
- The What and How of Work Functions
- Surface Science Definitions
- Work Function "Myths"
- Manipulating Work Functions
- Motivation for New TCOs
- Amorphous TCOs
- What the Future Holds

#### What About p-Type TCOs?

# The Strategy of Kawazoe et al.

- Delocalize O-2p levels at the VBM through enhanced covalency
  - Quasi-tetrahedral coordination of oxygen by cations
  - Linear coordination of cations
- Chose d<sup>10</sup> cations with energy levels close to that of O-2p





H. Kawazoe, H. Yanagi, K. Ueda, and H. Hosono. MRS Bull. 25(8)28 (2000).

# Linear cation TCO family



- Alternating layers of octahedra and Cu in linear coordination
- Two stacking sequences: rhombohedral and hexagonal
- p-type!

H. Kawazoe, H. Yanagi, K. Ueda, and H. Hosono. MRS Bull. 25(8)28 (2000).

### Band structure of CuAlO<sub>2</sub>



B.J. Ingram, T.O. Mason, R. Asahi, K.T. Park, and A.J. Freeman Phys. Rev. B. 64, 155114, (2001).

### CuAlO<sub>2</sub> Single Crystal Electrical Results

Band Conductor

- Relatively low, anisotropic mobilities
  - *ab*-plane: 3.0 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>
  - -c-direction: 0.12 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>

• Tough to dope!

Tate, J. et al. Phys. Rev. B. 80, 165206, (2009).



#### Review of literature data for delafossites

	σ (S/cm)	%T (visible) (film thickness)	E <sub>g</sub> (eV)
CuYO <sub>2</sub> :Ca <sup>(1)</sup>	1.0	40-50 (250 nm)	3.5
CuScO <sub>2</sub> <sup>(2)</sup>	15-30	40 (110 nm)	
CuGaO <sub>2</sub> <sup>(3)</sup>	6.3x10 <sup>-2</sup>	80 (500 nm)	3.6
CulnO <sub>2</sub> :Ca <sup>(4)</sup> CulnO <sub>2</sub> :Sn	2.8x10 <sup>-3</sup> (p) 3.8x10 <sup>-3</sup> (n)		3.9
CuAIO <sub>2</sub> <sup>(5)</sup>	16	32-52 (120 nm)	3.6

- 1. M. K. Jayaraj, et al. *Thin Solid Films*. **397**, 244 (2001).
- 2. N. Duan, et al. *Appl. Phys. Lett.* **77**(9), 1 (2000).
- 3. K. Ueda, et al. J. Appl. Phys. **89**(3), 1790 (2001).
- 4. H. Yanagi, et al. Appl. Phys. Lett. **78**(11), 1583 (2001).
- 5. Y. Wang, et al. *Chem. Vap. Deposition.* **6**(6), 285 (2000).

A New Approach: Non-d<sup>10</sup> Cations

# d<sup>6</sup> p-TCOs: Spinels



Vertices of each polyhedron are oxygen anions.

Normal II-III Spinels:

 $O_h$  site (e.g., Rh, Cr, Co)  $A^{+II} B_2^{+III} O_4$  J $T_d$  site (e.g., Zn, Mg, Mn)

### d<sup>6</sup> p-TCOs: Spinels



Crystal field splitting gives rise to transparency.

# Transmission Data for $ZnM_2O_4$ (M = Ir, Rh, Co)



M. Dekkers et al., Appl Phys. Lett. 90, 021903, (2007).

# Electrical Data for d<sup>6</sup> Spinels

Composition	σ (S/cm)	Eg
ZnIr <sub>2</sub> O <sub>4</sub>	3.39	2.97
$ZnRh_2O_4^{(1)}$	2.75	2.74
ZnCo <sub>2</sub> O <sub>4</sub> <sup>(1)</sup>	0.39	2.26
ZnCo <sub>2</sub> O <sub>4</sub> :Mg <sup>(2)</sup> (bulk)	0.9	

#### Low conductivities attributed to small hole mobilities!

(1) M. Dekkers *et al.*, Appl Phys. Lett. **90**, 021903, (2007).
(2) J. Perkins *et al.*, Phys. Rev. B. **84**, 205207, (2011).

#### Hall Effect: 17 T and 0.58 T





### Hall mobility at room temperature



A. Nagaraja et al., J. Am. Ceram. Soc. (2011) doi: 10.1111/j.1551-2916.2011.04771.x

#### Zn-Co-O is a Good Hole Transport Layer (HTL) for OPV



In collaboration with A. Sigdel, N.E. Widjonarko and J.J. Berry – CIS:SEM EFRC

#### Performance comparable to PEDOT: PSS on the first try!

Back to n-Type TCOs: The Importance of Work Functions

# The Need for Ohmic Contacts

# **Organic Photovoltaic (OPV)** $\Theta \Theta \Theta$ Photons $\oplus$ $\oplus$ Cathode Anode (TCO) Active Layer (organic)

- TCOs play an integral role as electrically transparent contacts in solar cells
- Device performance depends on Ohmic contacts with electrodes
- Contact type depends on band alignment between layers

# What is the Work Function? How is it Measured?

### Photoelectron Spectroscopy



#### **XPS Survey Spectra**



c-ZITO-10



### In Situ Photoelectron Spectroscopy



Technical University of Darmstadt (Darmstadt, Germany)

#### Kelvin Probe WF Measurements



#### Kelvin Probe Method



Under ambient conditions!



Harvey et al., J. Am. Ceram. Soc. 91 11 (2008)



Harvey et al., J. Am. Ceram. Soc. 91 11 (2008)

#### **Comparison of KP and UPS**



#### Some Surface Science Definitions

#### Some Basic Definitions



- $E_{vac}$  = vacuum level
- E<sub>F</sub> = Fermi level
- $E_{CBM}$  = conduction band minimum

 $E_{VBM}$  = valence band maximum

- $E_{g0}$  = fundamental band gap
- $\Delta E_g$  = Burstein-Moss shift
- $I_p$  = ionization potential (E<sub>F</sub> – E<sub>VBM</sub>) = Fermi level position WF = work function (Φ)

#### **Debunking Work Function Myths**

# **TCO Work Function Myths**

- Every material has a unique work function
- The work function of a given oxide's surface is a constant
- The only way to change the work function of an oxide surface is "extrinsically" (e.g., by surface adsorbates)\*



\* For example, Neal Armstrong et al., Accts. Chem. Res., 42 (11) 1748 (2009).

#### Manipulating Work Functions

# How to Modify Work Function?

- Two ways:
  - Shift the Fermi level position

 $(E_F - E_{VBM})$  can change independently with doping

 Change the Ionization potential

 $(E_{vac}-E_{VBM})$ 

can change independently with surface dipole changes (assuming E<sub>g</sub> is constant)



#### **Ionization Potential Plots**



$$I_p = (E_F - E_{VBM}) + WF(\Phi) \rightarrow WF(\Phi) = I_p - (E_F - E_{VBM})$$

### **Changing Fermi Level Position**

(ITO) 6.0 ITO Energy 5.5  $\mathsf{E}_{\mathsf{vac}}$ work function [eV] 5.0-🕱 I<sub>P</sub> = 8.1 eV  $\boxtimes$ 4.5l<sub>P</sub> = 7.6 eV 4.0-CB I<sub>P</sub> = 7.0 eV E<sub>F</sub> ⁻ŗ 3.5-Sn defect band E<sub>F</sub> • E<sub>g0</sub>, 4.0 2.0 2.5 3.0 3.5 E<sub>F</sub>-E<sub>VB</sub> [eV]

A. Klein et al., Materials, 3 (2010) 4892

 $IP = (E_F - E_{VB}) + WF$ 

VB

**Burstein-Moss Shift** 

# **ITO Defect Chemistry** ITO $n = 2[V_0'] n = [Sn_{ln}] n = [Sn_{ln}]$ log [] -1/6 -1/8 -(2Sn<sub>In</sub> O<sub>i</sub>")<sup>x</sup>

 $\log pO_2$ 

# How to Modify Work Function?

- Two ways:
  - Shift the Fermi level position

 $(E_F - E_{VBM})$  can change independently with doping

 Change the Ionization potential

 $(E_{vac}-E_{VBM})$ 

can change independently with surface dipole changes (assuming E<sub>g</sub> is constant)



# Surface Orientation Effects

#### Silver

Gold

Surface	Work Function	Polycrystalline Work Function	Surface	Work Function	Polycrystalline Work Function
Ag (111)	4.74 eV		Au (111)	5.31 eV	
Ag (100)	4.64 eV		Au (100)	5.47 eV	
Ag (110)	4.52 eV		Au (110)	5.37 eV	
Au(poly)		4.26 eV	Au(poly)		5.10 eV

Note: close-packed surfaces have high work functions; polycrystalline specimens represent averages with low-work function orientations.

H. B. Michaelson, The Work Function of the Elements and Its Periodicity, J. Appl. Phys. 48 (1977) 4729.

#### **ZnO-Surface Orientations**

ZnO has polar and non-polar surface terminations



# ZnO:Al Defect Chemistry



S. Lany, A. Zunger, PRL, 98, 045501 (2007)

# SnO<sub>2</sub> Surface Terminations

SnO<sub>2</sub>

3.5

4.0

3.0

E<sub>F</sub>-E<sub>VB</sub> [eV]



Green = Bulk Sb:SnO<sub>2</sub>

M. Batzill et al., Phys. Rev. B, **72** (2005) 165414

# SnO<sub>2</sub>:Sb Defect Chemistry



# **TCO Work Function Truths**

- The WF of an oxide surface decreases linearly as Fermi level increases (if IP remains constant)
- The IP (WF) of an oxide surface can be changed by dipole modifications—surface orientation, surface termination, surface stoichiometry—(if Fermi level remains constant)
- We can therefore change oxide surface work functions both "intrinsically" and "extrinsically"\*

 $\mathsf{IP} = (\mathsf{E}_{\mathsf{F}} - \mathsf{E}_{\mathsf{VBM}}) + \mathsf{WF}$ 



\* For example, Neal Armstrong et al., Accts. Chem. Res., 42 (11) 1748 (2009).

#### Motivation for New TCOs

# Why Do We Need New TCOs?

- Why do people use ITO?
  - Available
  - Good TCO (up to 10,000 S/cm conductivity, good transparency)
- Drawbacks to ITO
  - COST
    - In costs are volatile
  - STABILITY
    - ITO is etched by common barrier layer PEDOT:PSS
  - WORK FUNCTION
    - ITO work function is generally < 5eV



# The Importance of TCOs in OPVs

#### **Organic Photovoltaic (OPV)**



- TCOs play an integral role as electrically transparent contacts in solar cells
- Device performance depends on Ohmic contacts with electrodes
- Contact type depends on band alignment between layers

# TCOs in Traditional/Inverted Cells

#### **Traditional Polarity**

#### **Inverted Polarity**



Want high work function: > 5eV

Want low work function: < 4eV

Both: Want high Fermi level (high conductivity)

J. Nelson, Materials Today, 14 (2011) 462

### The Zn-In-Sn-O System



# ZITO (Zn,Sn co-substituted ITO) Surfaces



A. Klein et al., Thin Solid Films, 518 (2009) 1197

#### **Two-Step Processing**



# C-ZITO Bulk Structure: Bixbyite



#### bixbyite = fluorite with 1/4 of the anions missing

Two different cation positions Structural oxygen interstitial positions



G. Frank and H. Kostlin. Appl. Phys. A **27**, (1982) 197 Gonzalez and Mason, et al. J.A.P. **96**(7) (2004) 3912



- High  $\mu_0$  can produce a high  $\ln_2O_3$  work function on a peroxidized (100) surface)
- Polycrystalline materials will exhibit inhomogeneous surface work functions: "hot spots," "cold spots"

M.V. Hohmann et al., J. Phys.: Condens. Matter 23 (2011) 334203

# **Predicted Ionization Potentials**



- Ionization potentials
  - Vary little on (111) and (110) surfaces
  - Vary greatly on (100) surface
    - Metallic termination = low ionization potential
    - Oxide termination = high ionization potential

M.V. Hohmann et al., J. Phys.: Condens. Matter 23 (2011) 334203



A. Klein et al., Materials, 3 (2010) 4892



A. Klein et al., Materials, 3 (2010) 4892

#### **Two-Step Processing**



### Parallelogram Plots



#### Amorphous TCOs

# Advantages of Amorphous TCOs

- Easier, cheaper processing
  - Low temperature
  - Flexible substrates
  - Better etching characteristics
- Smoother surfaces
- Wide composition range
  - Minimization of toxic/expensive elements
- No hot/cold spots





Harvey et al., J. Am. Ceram. Soc. 91 11 (2008)

# Why a-TCOs Have Good Properties



Hosono, H., J. Noncryst. Sol. 352, 851-858 (2006)

#### **Pulsed Laser Deposited a-ZITO Films**



In2O3: No cosubstitution ZITO-30: 30 at.% cosubstituted ZITO-50: 50 at.% cosubstituted ZITO-70: 70 at.% cosubstituted



#### How a-ZITO Compares



D. E. Proffit et al., Thin Solid Films (submitted)



#### What the Future Holds

#### **Pulsed Laser Deposited a-ZITO Films**



In2O3: No cosubstitution ZITO-30: 30 at.% cosubstituted ZITO-50: 50 at.% cosubstituted ZITO-70: 70 at.% cosubstituted



#### Application of ZITO TCO & TOS for FETs

TOS ZITO-30 used for Field Effect Transistor (FET) Channel Layer



 $C_{i}$  for 300 nm SiO<sub>2</sub>: ~10 nF/cm<sup>2</sup>  $V_{G}$  and  $V_{DS}$ : ~100 V  $\mu_{FE}$ : ~10 cm<sup>2</sup>/V·s  $I_{on}$ :  $I_{off}$ : ~10<sup>6</sup> VT: ~18 V SS: ~5.2 V/decade Air and light stable

 $V_{\rm G}$  and  $V_{\rm DS} \sim 4.0$  V  $\mu_{\rm FE} \sim 20$  cm<sup>2</sup>/V·s  $I_{\rm on}$ : $I_{\rm off} \sim 10^5$   $V_{\rm T} \sim 1.2$  V SS ~0.26 V/decade Good Mechanic Flexibility *Transparency* ~75%



TCO ZITO-30 used for Gate, Source & Drain TOS ZITO-30 used for Channel Layer

On plastic substrate to form all amorphous, flexible, transparent FET



ZITO Films Grown at Room Temperature by PLD

# Conclusions

- TCO work functions can be modified "intrinsically" through:
  - Fermi level shifts (via donor doping—defect chemistry plays a vital role!)
  - Surface dipole modifications by:
    - Changing surface orientations (e.g., Al-doped ZnO)
    - Altering surface terminations (e.g., Sb-doped SnO<sub>2</sub>)
    - By changing surface stoichiometry (e.g., peroxidation, Sndoped In<sub>2</sub>O<sub>3</sub> or ITO, Zn,Sn-cosubstituted In<sub>2</sub>O<sub>3</sub> or ZITO)

# Conclusions (cont'd)

- Work function vs. Fermi level "parallelogram plots" allow comparison of candidate TCOs for OPV applications
  - AZO (Al-doped ZnO) or alternatives (Ga-doped ZnO) are best for low work function applications, e.g., in inverted OPV designs
  - ZITO (Zn-In-Sn-O) or ATO (Sb-doped SnO<sub>2</sub>) and alternatives (FTO, F-doped SnO<sub>2</sub>) are best for high work function applications, e.g., in traditional OPV designs
- Amorphous TCOs exhibit work functions and Fermi levels comparable to their crystalline counterparts (e.g., ZITO)
- An emerging "frontier" for TCOs is as Amorphous Oxide Semiconductors (AOSs)