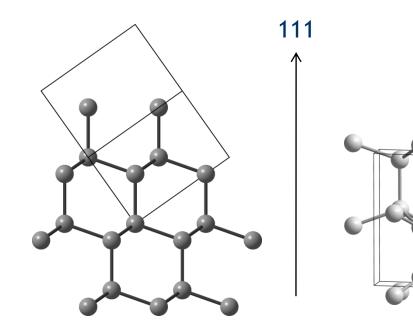


Diamond

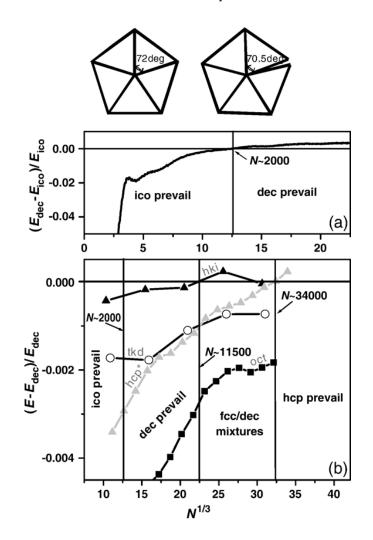
Wurtzite and zinc blende *P63mc* and *R3m/F-43m* 2*H* and 3*R*



Diamond

Lonsdaleite

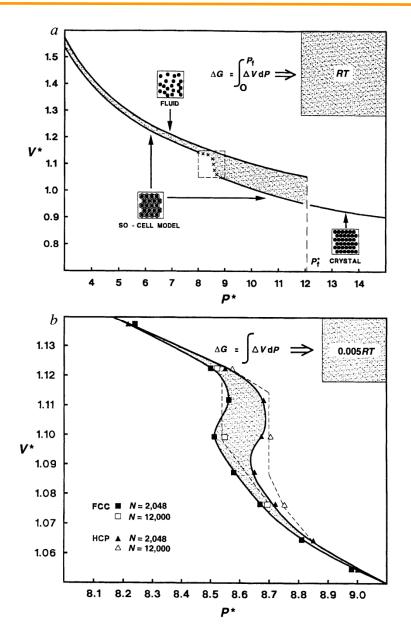
The *hcp-fcc* dilemma: Why is *fcc* more frequent than *hcp*? [25% of elements are *fcc* and 20% are *hcp*.



N. V. Krainyukova, The crystal structure problem' in noble gas nanoclusters, *Thin Solid Films*, 515 (2006) 1658-1663.

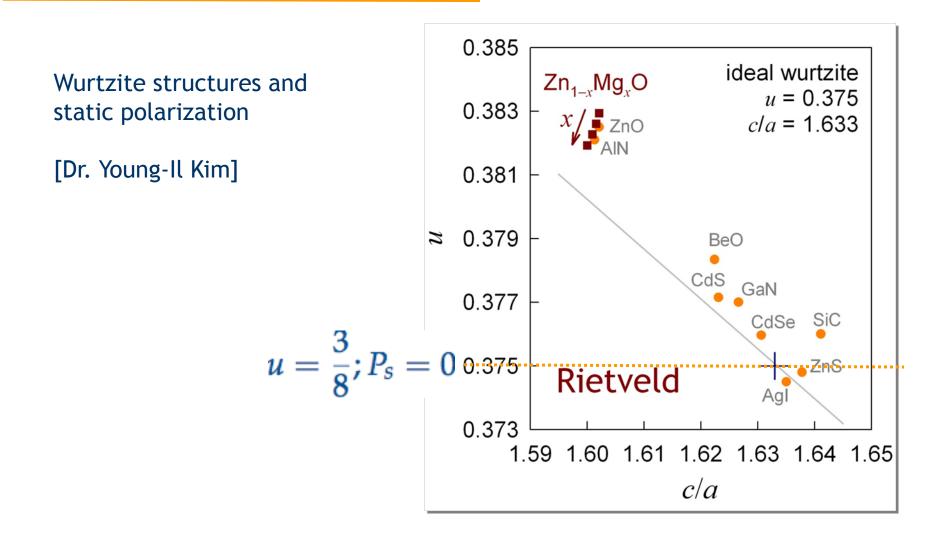
Calculations of the energetics of multiply twinned particles (MTPs) such as icosahedra and decahedra with fivefold symmetry as well as facecentered cubic (fcc) and hexagonal close-packed (hcp) particles in the size interval from 13 up to \sim 45,000 atoms were made applying Lennard-Jones potentials.

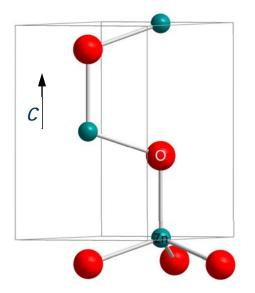
For the cluster size N from minimal up to N~2000 atoms the binding energy is highest for icosahedra, in the size interval from 2000 up to ~11,500 atoms decahedra prevail, above N~11,500 atoms decahedra and optimized fcc clusters were found to alternate. The hcp structure was revealed to become favorable above N~34,000 atoms. Thus, hcp clusters can attain their preference with respect to MTPs (comprising fcc fragments) and optimized fcc clusters only for very large sizes



Hard spheres: *fcc* is slightly stabilized by entropy over *hcp*, by 0.005 R for al temperatures up to the melting point.

L. V. Woodcock, Nature 385 (1997) 141.





Consider one ZnO₄ tetrahedron (polarization unit) Zn $(\frac{1}{3}, \frac{2}{3}, 0)$ O_{axial} $(\frac{1}{3}, \frac{2}{3}, u)$ O_{basal} $(\frac{2}{3}, \frac{1}{3}, u - \frac{1}{2})$ $(\frac{2}{3}, -\frac{2}{3}, u - \frac{1}{2})$ $(-\frac{1}{3}, \frac{1}{3}, u - \frac{1}{2})$

dipole moment along c-direction

$$\mu (\text{Zn-O}_{\text{axial}}) = \frac{1}{4} \times (-2) \times u \times c$$

$$\mu (\text{Zn-O}_{\text{basal}}) = 3 \times \frac{1}{4} \times (-2) \times (u - \frac{1}{2}) \times c \qquad (\text{in } e \cdot \text{\AA})$$

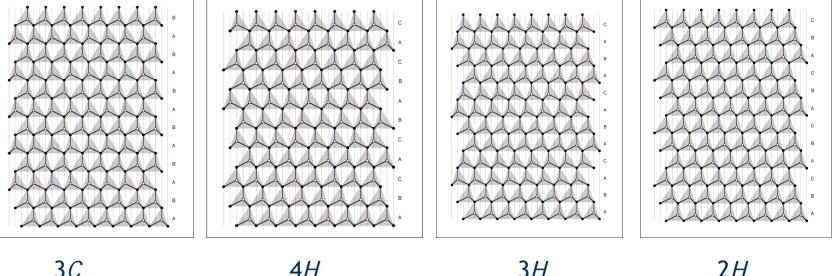
$$P_{s} = \frac{\mu_{sum}}{\text{unit volume}} = \frac{-2 \times \left(u - \frac{3}{8}\right) \times c}{\frac{\sqrt{3}}{4} a^{2} c} = \frac{-2 \times \left(u - \frac{3}{8}\right)}{\frac{\sqrt{3}}{4} a^{2}} \left(\frac{e}{\text{\AA}^{2}}\right) \times \left(1.60 \times 10^{3} \frac{\mu \text{C}}{\text{cm}^{2}} \cdot \frac{\text{\AA}^{2}}{e}\right)$$

 $(u - \frac{3}{8}) \cdot c$ can be obtained also from the separation between centers of (-) and (+) charges.

[Dr. Young-Il Kim]

Polytypism in SiC

Professor Marek Skowronski http://neon.mems.cmu.edu/skowronski/



3*C*

2H

Specifying the total number of layers in the hexagonal unit cell followed by the letter H, R or C to indicate the lattice type. Thus a symbol nH represents a structure with *n* layers in the primitive hexagonal unit cell while *mR* denotes a structure whose primitive lattice is rhombohedral and contains *m* layers in its hexagonal unit cell.

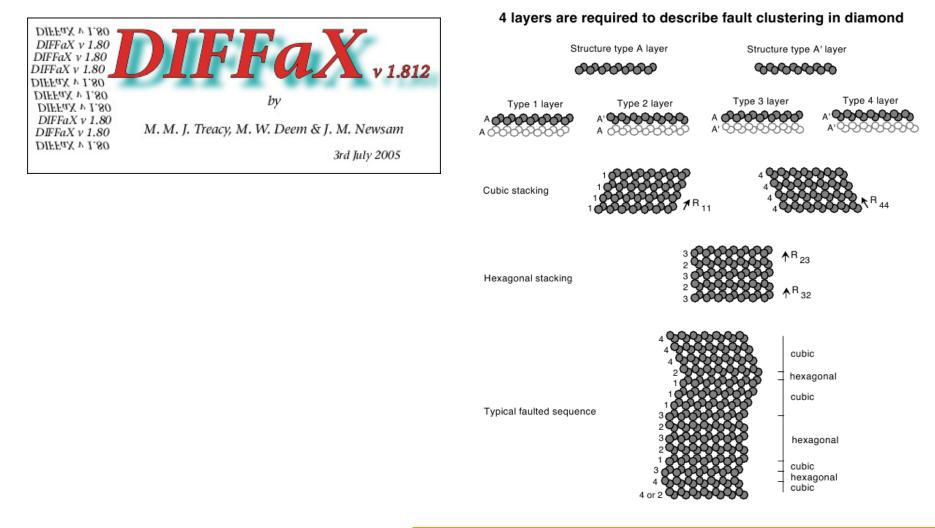
Ramsdell notation.

Stacking notation

Ramsdell notation	ABC sequence	Zhdanov number	h-c notation
2H	AB	(11)	h
3C	ABC		с
4H	ABCB	(22)	hc
6H ₁	ABCACB	(33)	hcc
6H ₂	ABCBAB	(2211)	hchchh
9R	ABACACBCB	(12)	hhc

http://www.iucr.org/iucr-top/comm/cteach/pamphlets/5/node0.html

DIFFaX: Diffraction from faulted structures:



Semiconductors: Structure sorting (Phillips-van Vechten)

